

# The SIESTA method for electronic structure calculations

Spanish

Initiative for

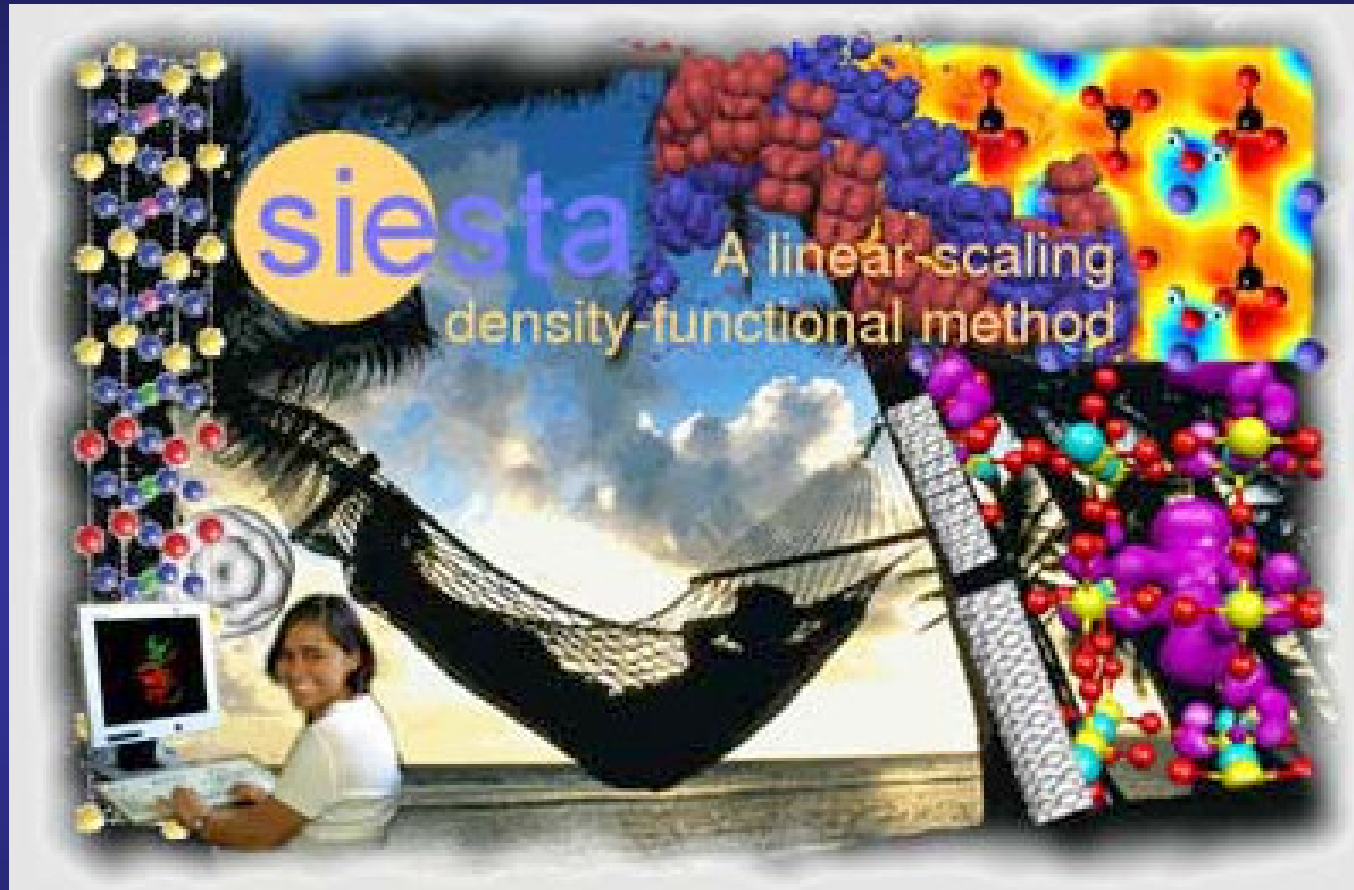
Electronic

Simulations with

Thousand of

Atoms

<http://www.uam.es/siesta>



Javier Junquera

UC

UNIVERSIDAD  
DE CANTABRIA

# What is SIESTA?

(<http://www.uam.es/siesta>)



Is both a **method** and its **computer program implementation** to simulate molecules and solids at the atomic scale.

Performs **electronic structure calculations**: **solves** numerically the **quantum mechanical equations** that rules the behaviour of the electrons

Performs **molecular dynamics** simulations of physical and chemical processes that occur at the atomic scale.

**EVERYTHING FROM FIRST-PRINCIPLES OR *AB-INITIO***

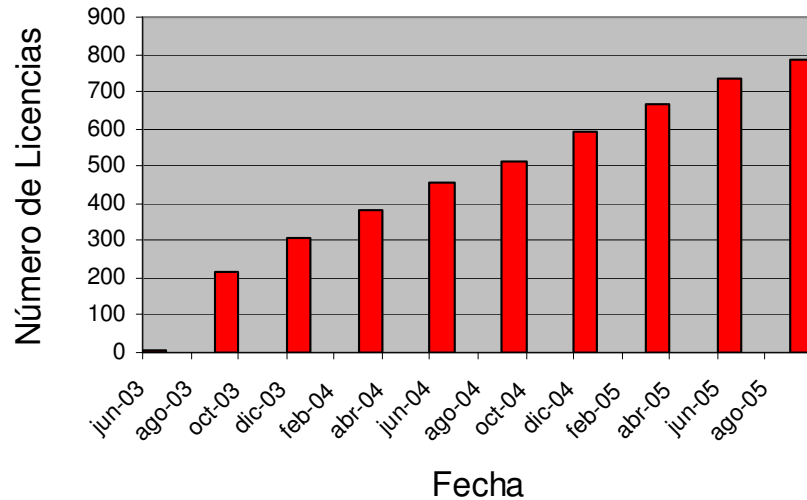
# Who are the members of the SIESTA core?

Collaboration between different Condensed Matter groups,  
with a strong Spanish contribution

<b>Emilio Artacho</b>	University of Cambridge (United Kingdom)
<b>Julian D. Gale</b>	Curtin University of Technology (Australia)
<b>Alberto García</b>	ICMAB (CSIC; Barcelona)
<b>Javier Junquera</b>	Universidad de Cantabria
<b>Richard M. Martin</b>	University of Illinois at Urbana-Champaign
<b>Pablo Ordejón</b>	ICMAB (CSIC; Barcelona)
<b>Daniel Sánchez-Portal</b>	DIPC (CSIC; San Sebastián)
<b>José M. Soler</b>	Universidad Autónoma de Madrid
<b>José M. Cela's group</b>	Barcelona Supercomputing Center

# International impact of SIESTA

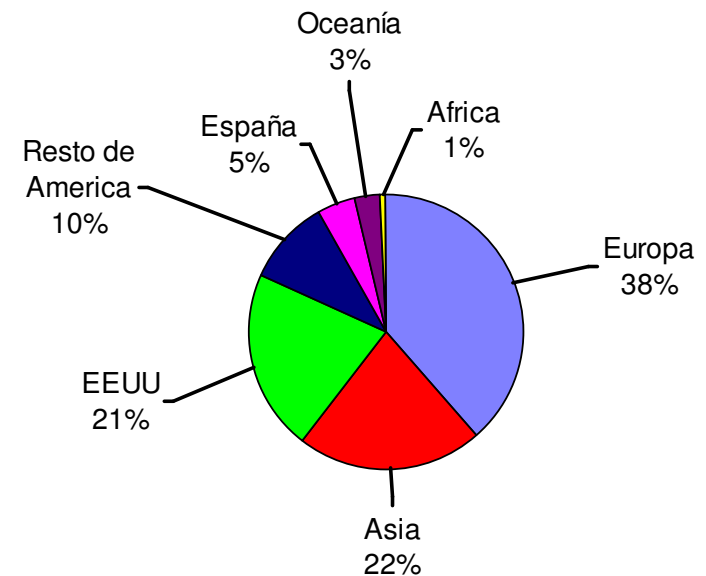
Figura 1: Licencias Académicas de SIESTA v. 1.3



Steadily growth of the number of users

Presence in the five continents

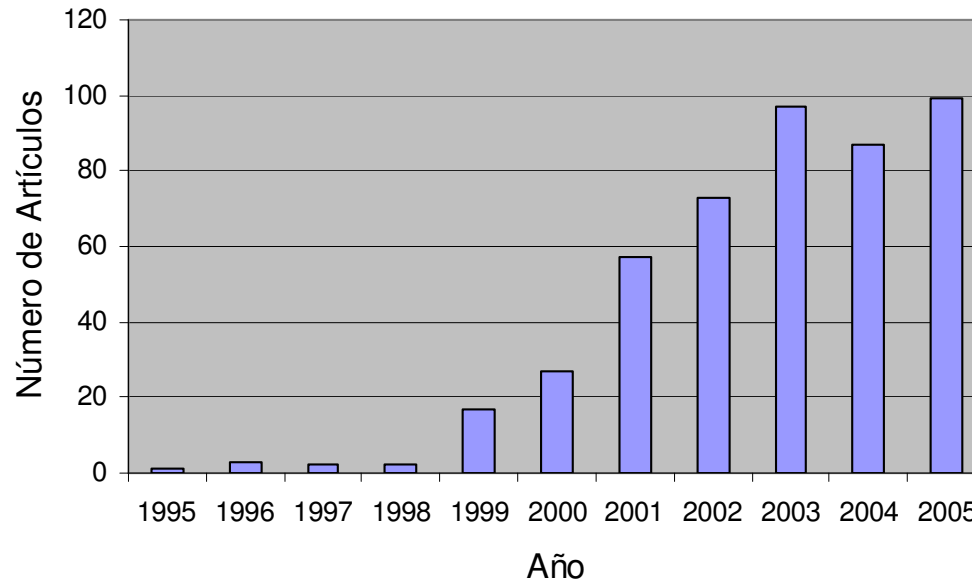
Figura 2: Distribución geográfica de Licencias





# More than 500 publications using the code

**Figura 3: Publicaciones realizadas con SIESTA  
(datos de 2005 sólo hasta Septiembre)**



Review	Impact factor	Number of papers
Nature	32,182	4
Science	31,853	2
Nature Materials	13,531	2
Phys. Rev. Lett.	7,218	46
Phys. Rev. B	3,075	168

# Most important reference followed in this lecture

INSTITUTE OF PHYSICS PUBLISHING

JOURNAL OF PHYSICS: CONDENSED MATTER

J. Phys.: Condens. Matter **14** (2002) 2745–2779

PII: S0953-8984(02)30737-9

## The SIESTA method for *ab initio* order- $N$ materials simulation

José M Soler<sup>1</sup>, Emilio Artacho<sup>2</sup>, Julian D Gale<sup>3</sup>, Alberto García<sup>4</sup>,  
Javier Junquera<sup>1,5</sup>, Pablo Ordejón<sup>6</sup> and Daniel Sánchez-Portal<sup>7</sup>

# Goal: Describe properties of matter from theoretical methods firmly rooted in fundamental equations

**Quantum Mechanics:** Schrödinger equation (assuming no relativistic)

$$i\hbar \frac{\partial \Psi \left( \{\vec{x}_i\}, \{\vec{R}_\alpha\}; t \right)}{\partial t} = \hat{H} \Psi \left( \{\vec{x}_i\}, \{\vec{R}_\alpha\}; t \right)$$

Coordinates of electron  $i$   $\vec{x}_i$  comprise

$$\begin{cases} \text{Space coordinates} & \vec{r}_i \\ \text{Spin coordinates} & \sigma_i \end{cases}$$

**Electromagnetism:** Coulomb's law

$$\hat{H} = \hat{T} + \hat{V}_{Coulomb}$$

For a pair of charged particles

$$\hat{V}_{Coulomb} = \frac{q_i q_j}{|\vec{r}_i - \vec{r}_j|}$$

# A closer look to the hamiltonian: A difficult interacting many-body system.

$$\hat{H} = \sum_i -\frac{\hbar^2}{2m_e} \nabla_i^2 - \sum_{i,\alpha} \frac{Z_\alpha e^2}{|\vec{r}_i - \vec{R}_\alpha|}$$

Kinetic energy operator for the electrons

Potential acting on the electrons  
due to the nuclei

$$+ \frac{1}{2} \sum_{i \neq j} \frac{e^2}{|\vec{r}_i - \vec{r}_j|}$$

Electron-electron interaction

$$+ \sum_\alpha -\frac{\hbar^2}{2M_\alpha} \nabla_\alpha^2$$

Kinetic energy operator for the nuclei

$$+ \frac{1}{2} \sum_{\alpha \neq \beta} \frac{Z_\alpha Z_\beta e^2}{|\vec{R}_\alpha - \vec{R}_\beta|}$$

Nucleus-nucleus interaction

# What are the main approximations?

## Born-Oppenheimer

Decouple the movement of the electrons and the nuclei.

## Density Functional Theory

Treatment of the electron – electron interactions.

## Pseudopotentials

Treatment of the (nuclei + core) – valence.

## Basis set

To expand the eigenstates of the hamiltonian.

## Numerical evaluation of matrix elements

Efficient and self-consistent computations of  $H$  and  $S$ .

## Solve the secular equation

## Supercells

To deal with periodic systems

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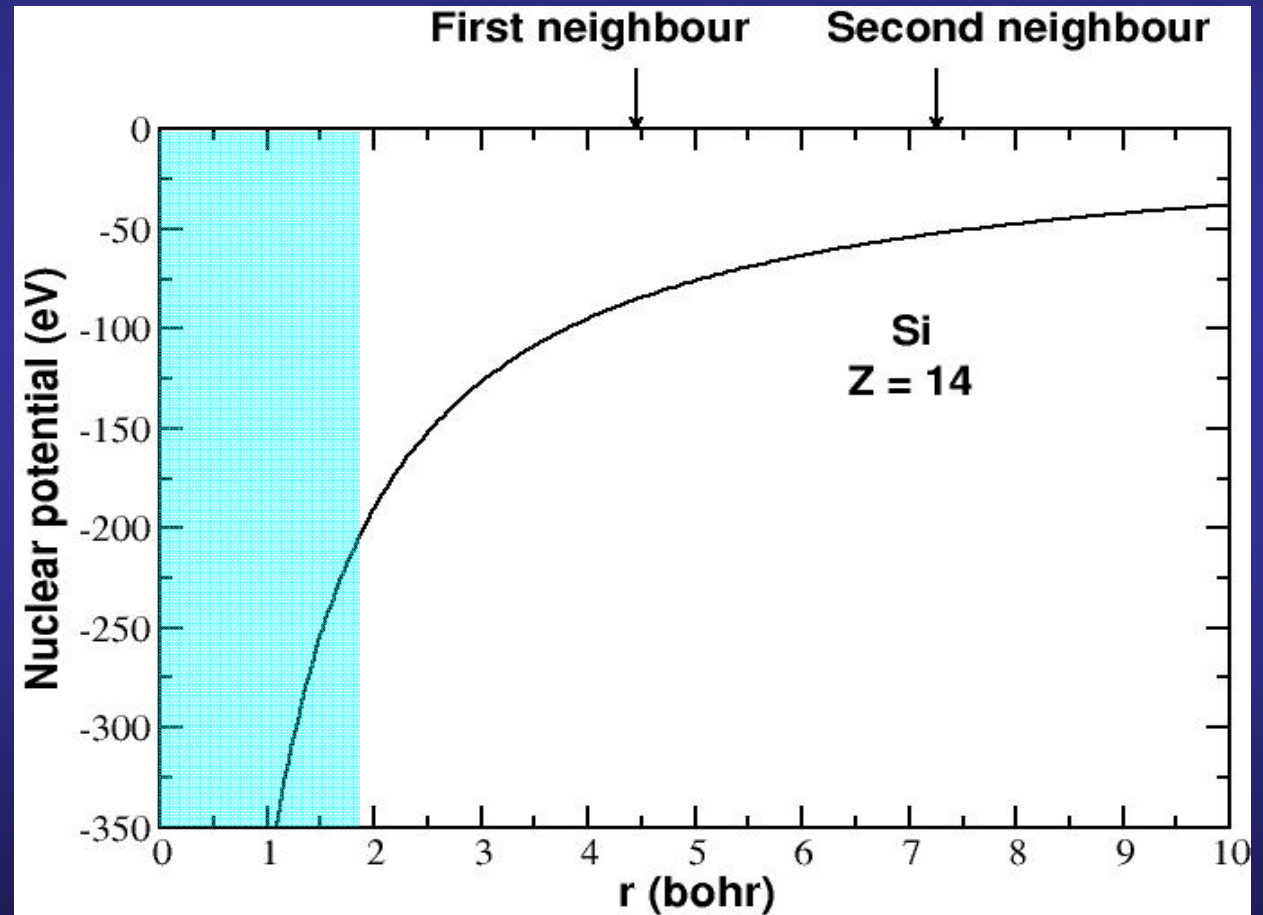
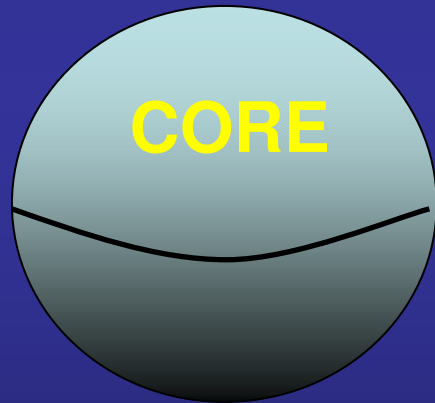
Efficient and self-consistent computations of  $H$  and  $S$ .

## Solve the secular equation

## Supercells

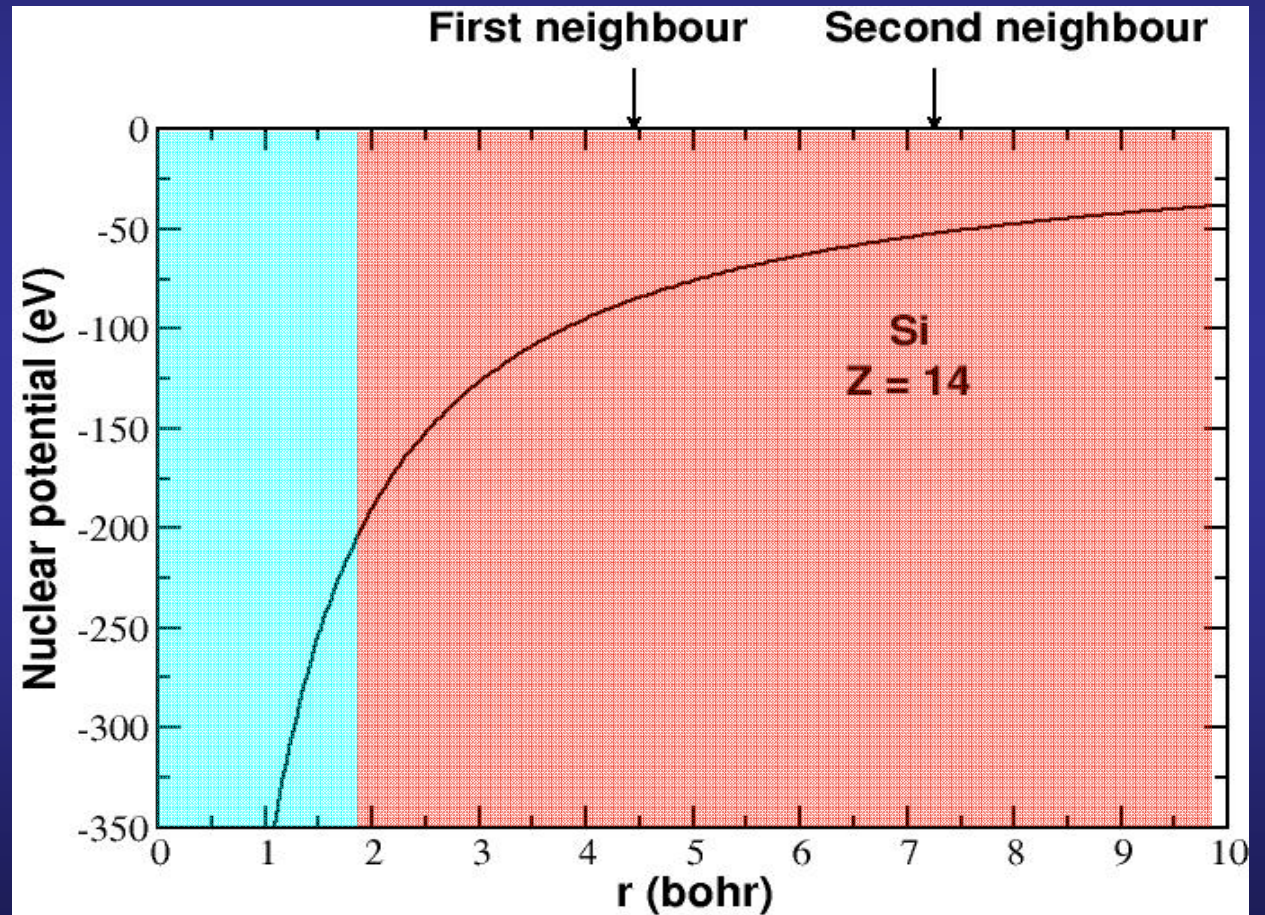
To deal with periodic systems

# Difficulty: how to deal accurately with both the core and valence electrons



**Difficulty: how to deal accurately with both the core and valence electrons**

**VALENCE**

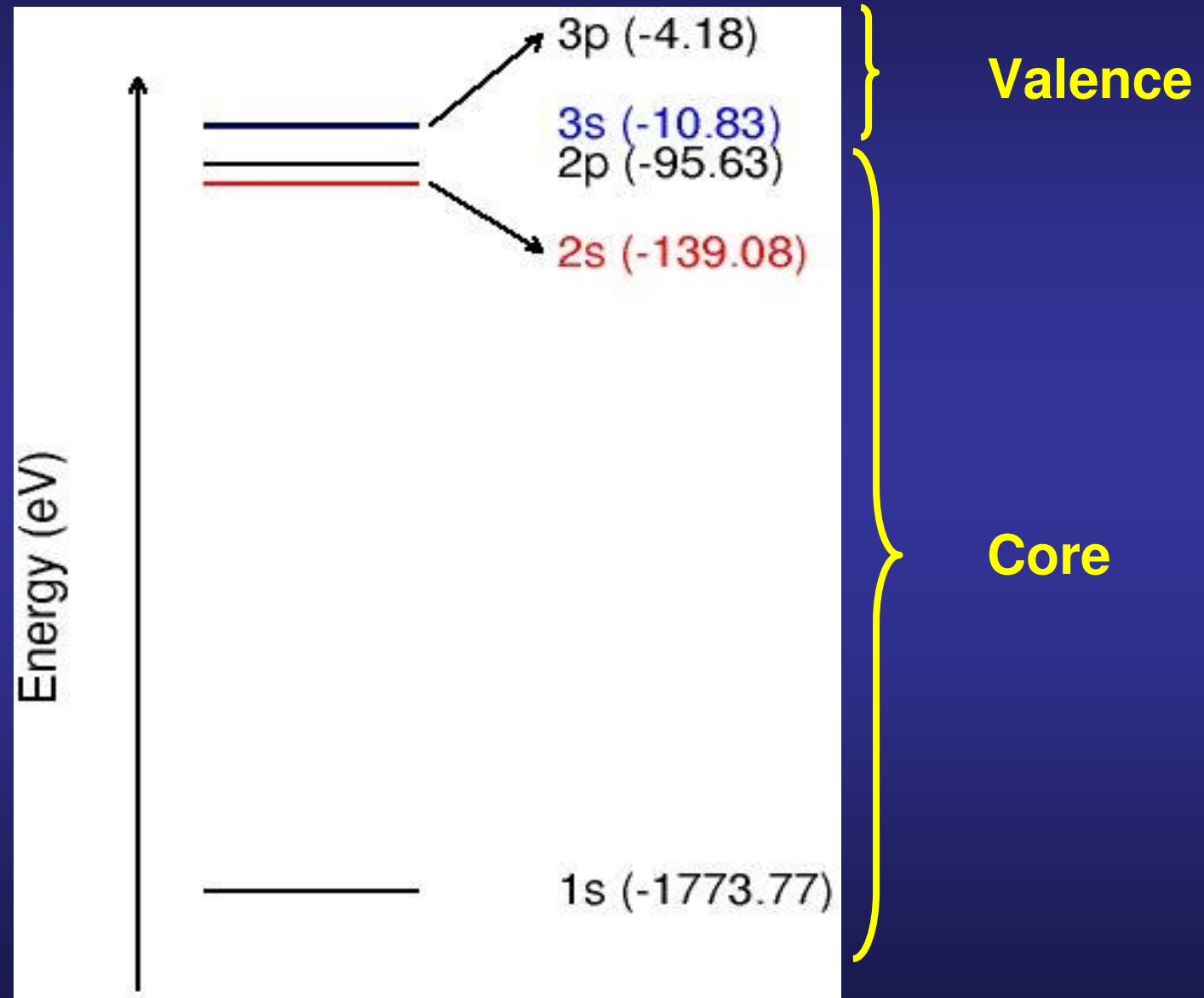




Si atomic configuration:  $\underbrace{1s^2 2s^2 2p^6}_{\text{core}} \underbrace{3s^2 3p^2}_{\text{valence}}$

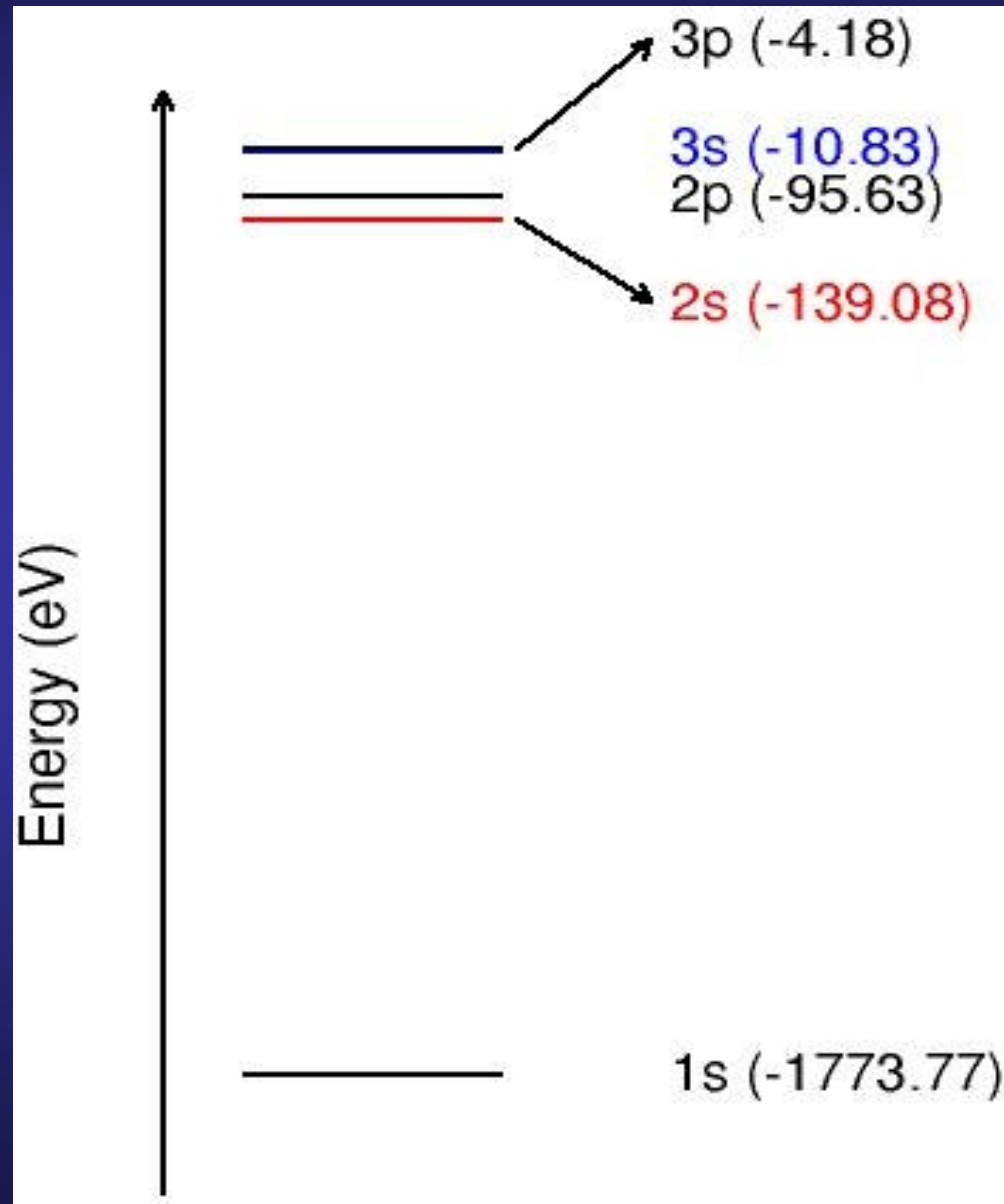
H																	He
Li	Be											B	C	N	O	F	Ne
Na	Mg											Al	Si	P	S	Cl	Ar
K	Ca	Sc	Ti	V	Cr	Mn	Fe	Co	Ni	Cu	Zn	Ga	Ge	As	Se	Br	Kr
Rb	Sr	Y	Zr	Nb	Mo	Tc	Ru	Rh	Pd	Ag	Cd	In	Sn	Sb	Te	I	Xe
Cs	Ba		Hf	Ta	W	Re	Os	Ir	Pt	Au	Hg	Tl	Pb	Bi	Po	At	Rn
Fr	Ra		Rf	Db	Sg	Bh	Hs	Mt	Uun	Uuu	Uub						
			La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb	Lu
			Ac	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No	Lr

# Core eigenvalues are much deeper than valence eigenvalues

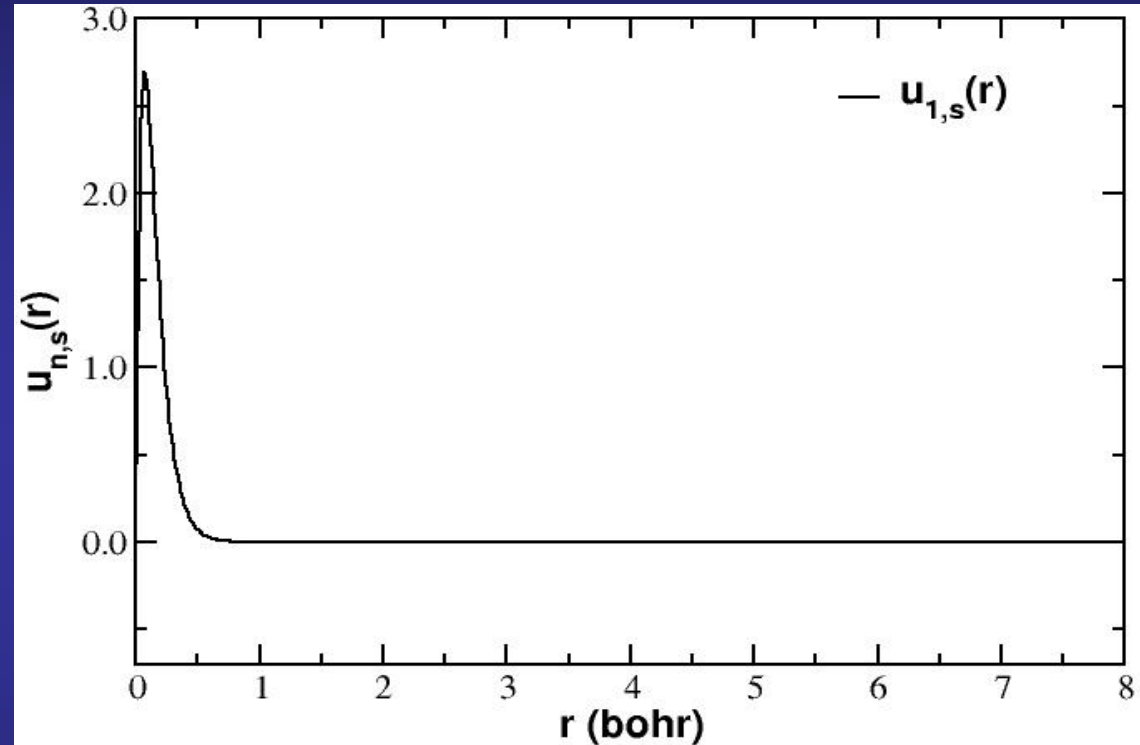


Atomic Si

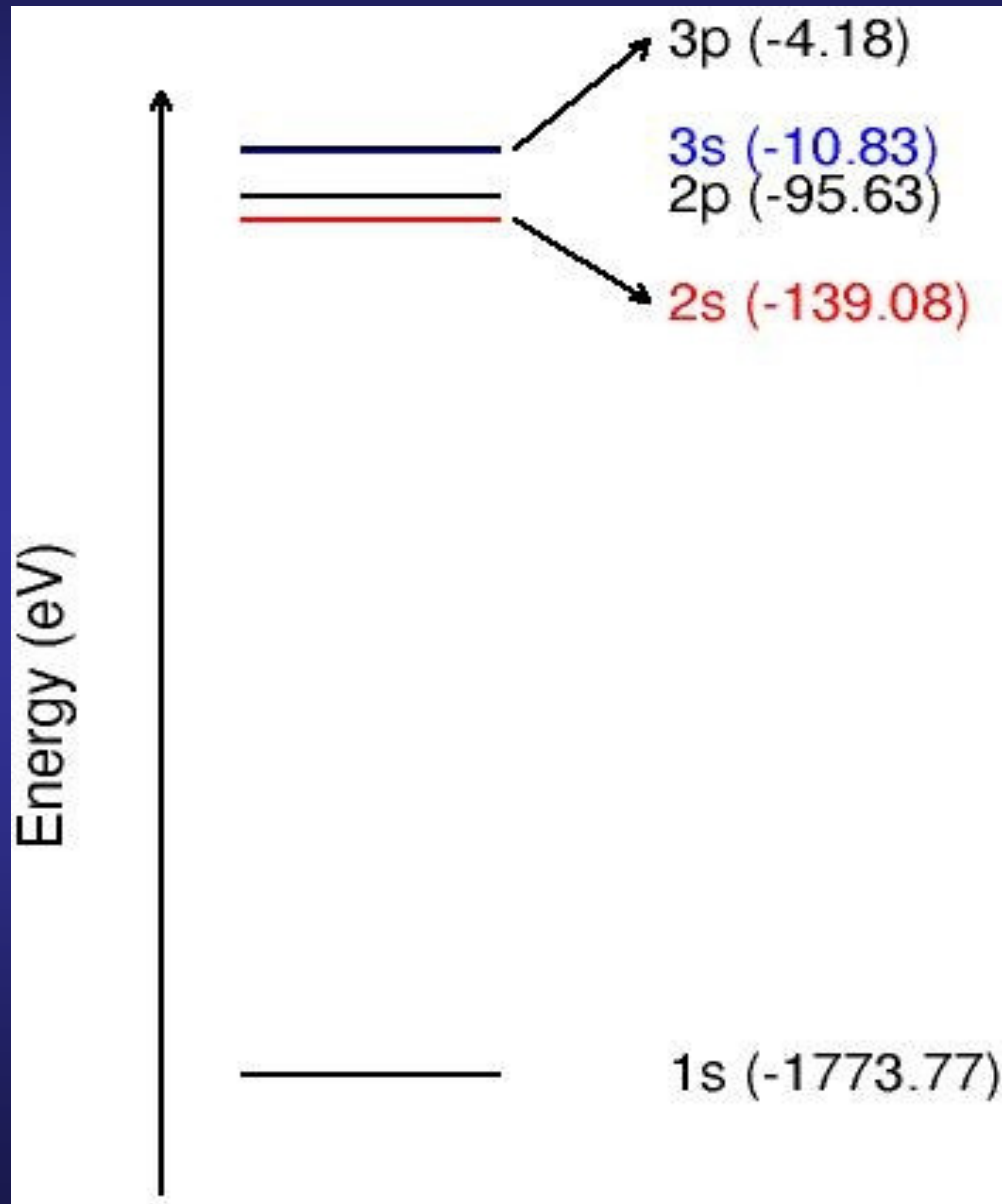
# Core wavefunctions are very localized around the nuclei



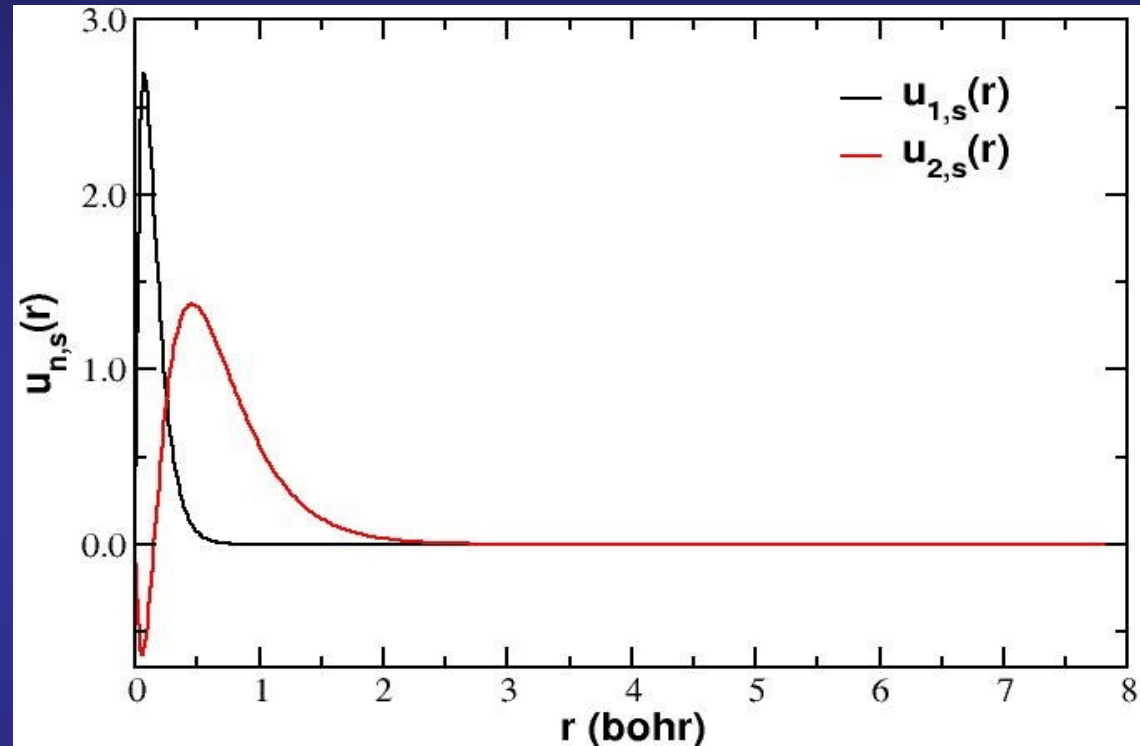
Atomic Si



# Core wavefunctions are very localized around the nuclei



Atomic Si



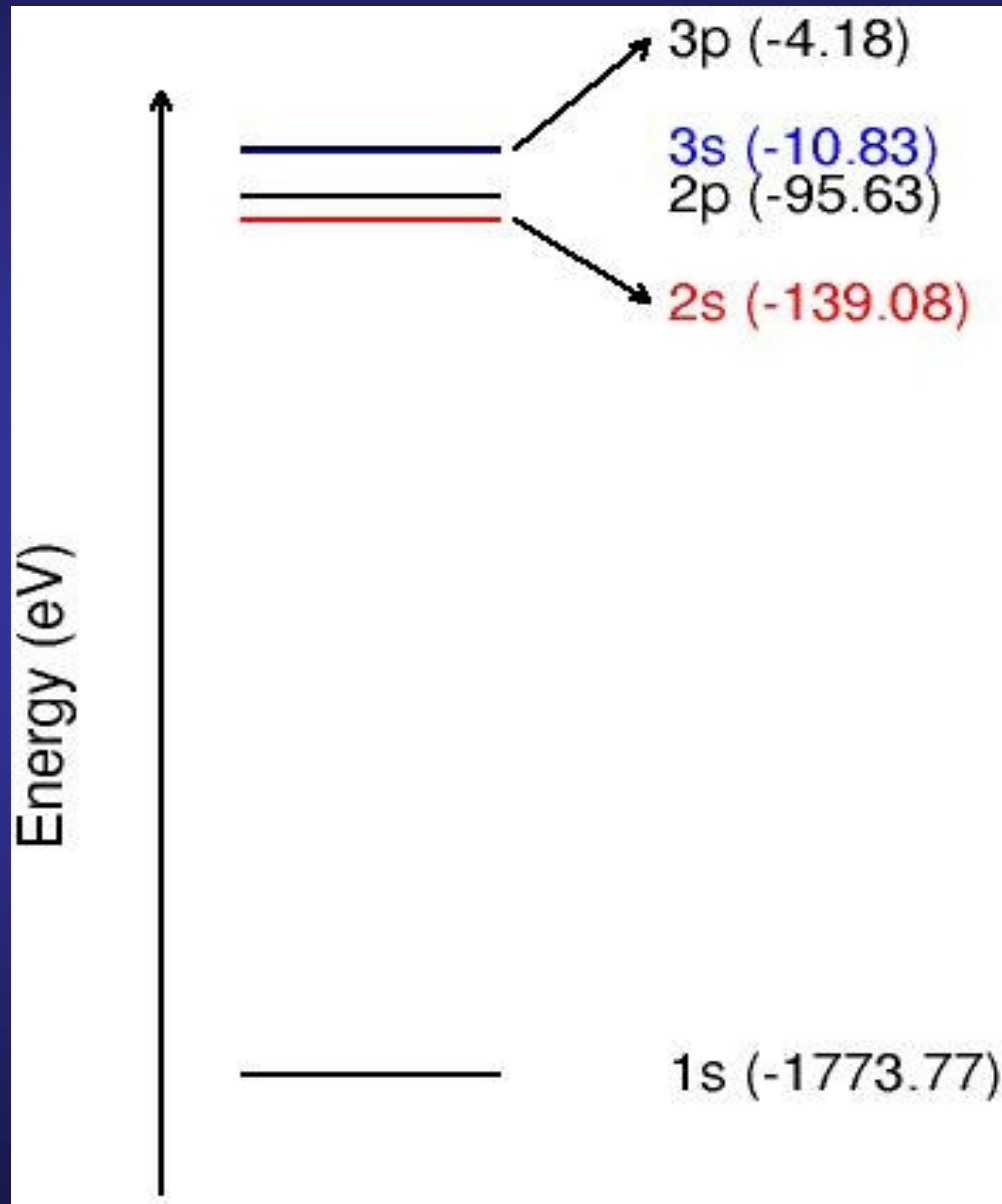
## Core electrons...

highly localized

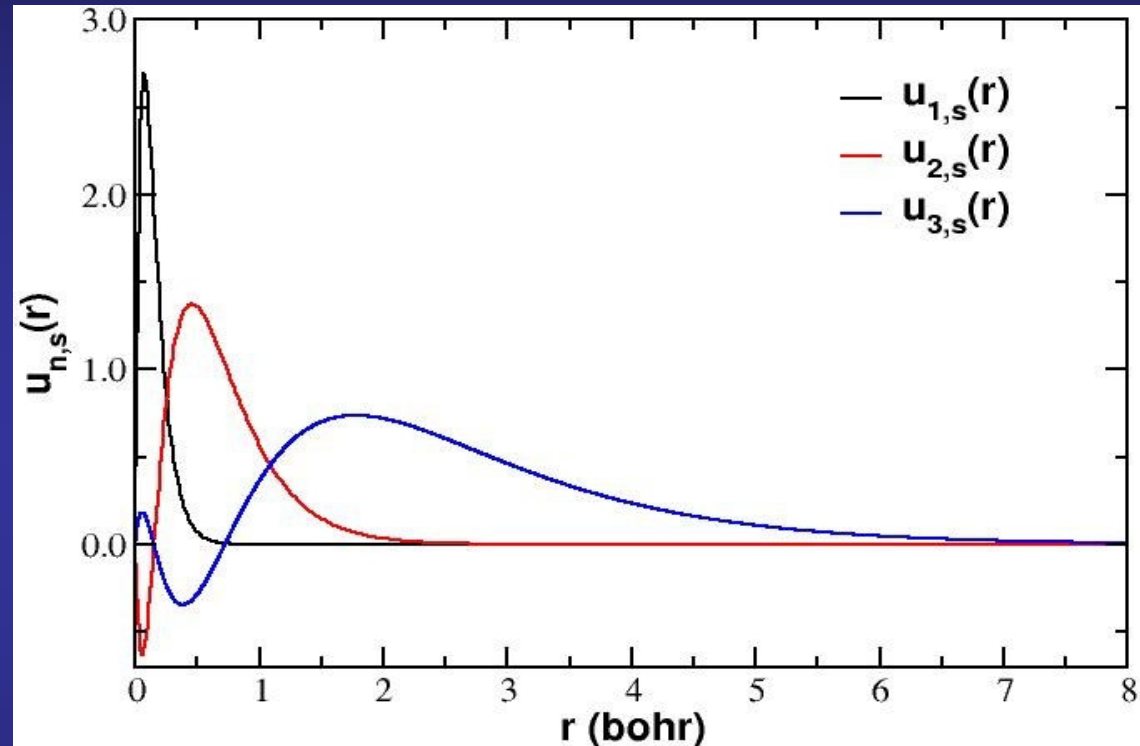
very depth energy

... are chemically inert

# Valence wave functions must be orthogonal to the core wave functions



Atomic Si



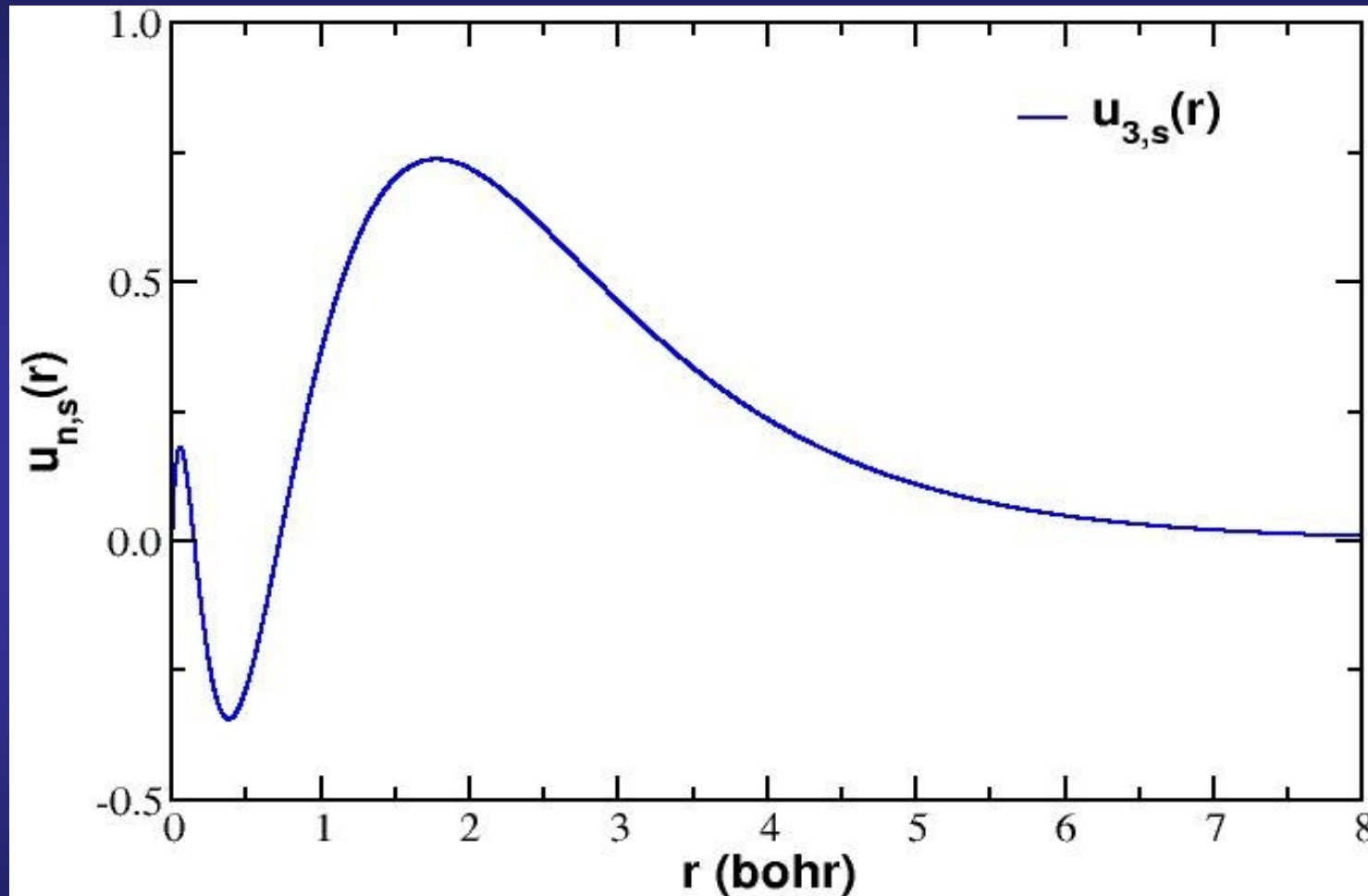
Core electrons...

highly localized

very depth energy

... are chemically inert

**Fourier expansion of a valence wave function has a great contribution of short-wave length**



**To get a good approximation we would have to use a large number of plane waves.**

# **Pseudopotential idea:**

**Core electrons are chemically inert**  
**(only valence electrons involved in bonding)**

**Core electrons make the calculation more expensive**

**more electrons to deal with**

**orthogonality with valence  $\Rightarrow$  poor convergence in PW**

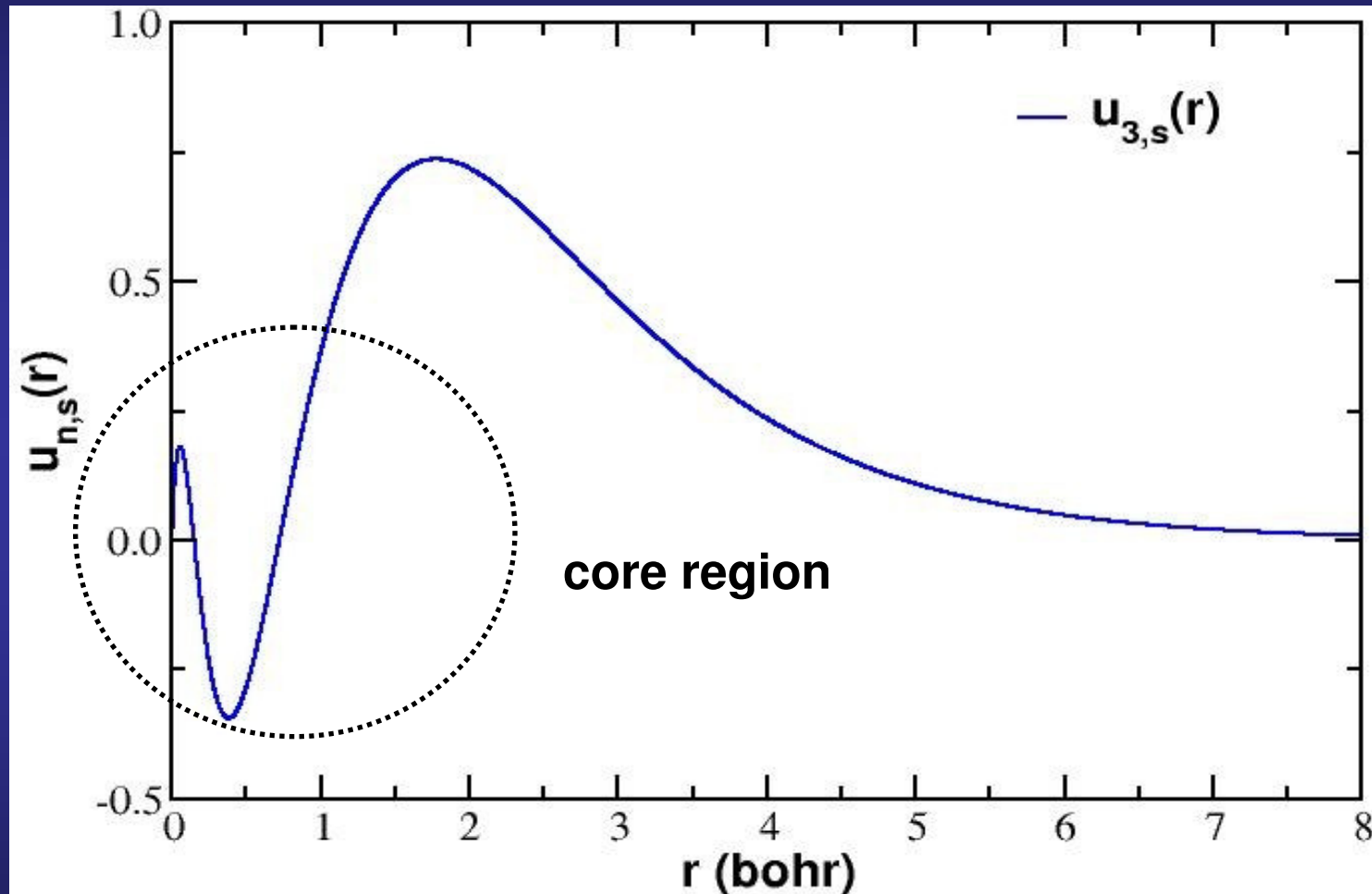
**Core electrons main effect: screen nuclear potential**

**Idea:**

**Ignore the dynamics of the core electrons (freeze them)**

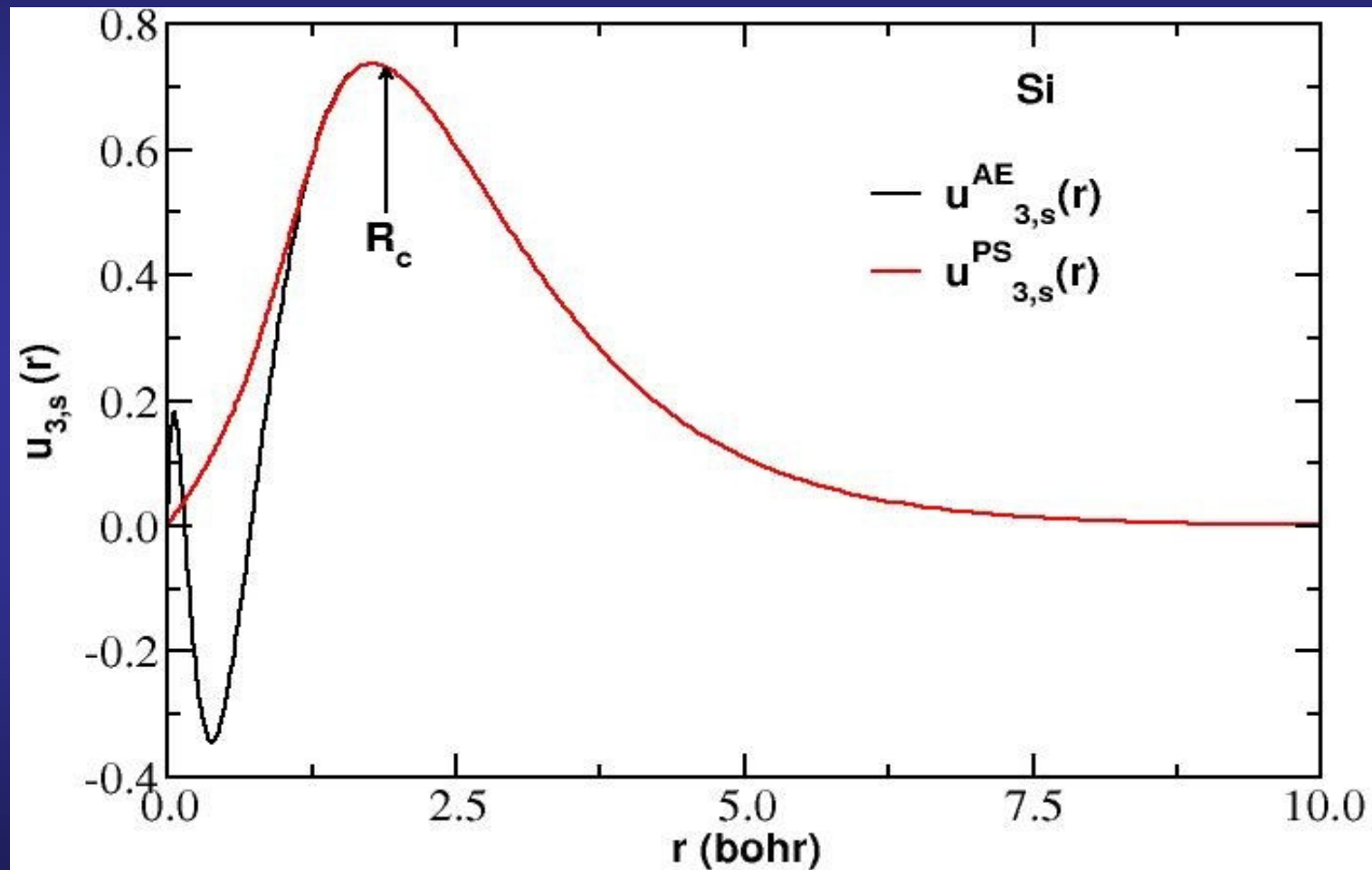
**And replace their effects by an effective potential**

The nodes are imposed by orthogonality to the core states

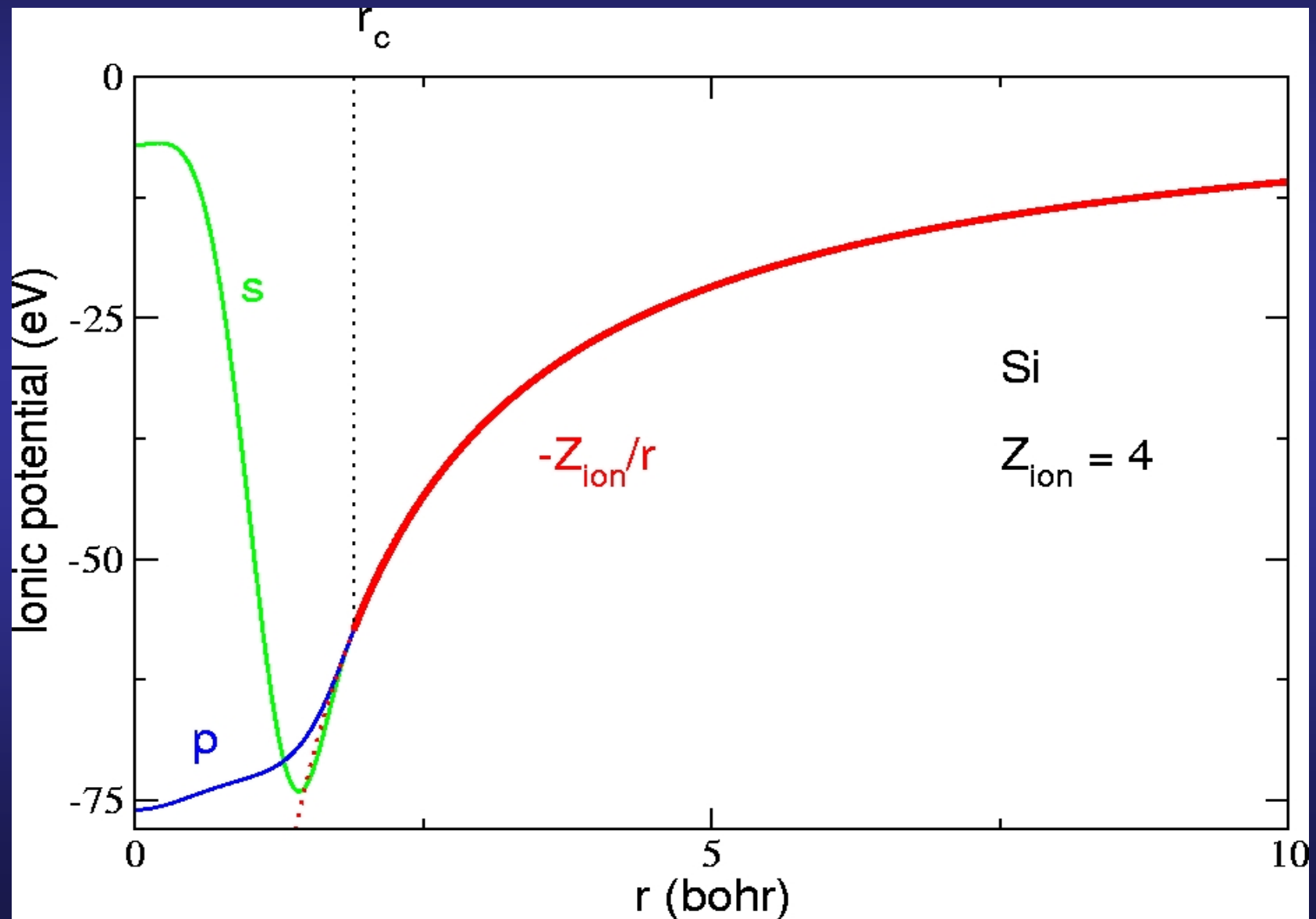




Idea, eliminate the core electrons by  
ironing out the nodes



***Ab-initio* pseudopotential method:  
fit the valence properties calculated from the atom**





# A good starting point: pseudopotentials tabulated in the SIESTA web page

SIESTA (Spanish Initiative for Electronic Simulations with Thousands of Atoms) - Mozilla Firefox

Archivo Editar Ver Historial Marcadores Herramientas Ayuda

http://www.uam.es/departamentos/ciencias/fismateriac/siesta/

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webmaster:  
siesta.web@uam.es

## Pseudopotential/Basis Database

This section contains links to pseudopotentials which were obtained from [ABINIT's Fritz-Haber-Institute \(FHI\) pseudo database](#)

The SIESTA team would like to thank the ABINIT team for sharing their pseudopotentials with the community.

[Translation of ABINIT's LDA pseudo database to SIESTA format](#)

[Translation of ABINIT's GGA pseudo database to SIESTA format](#)

### User contributed database

W3C HTML 4.01 W3C CSS

Terminado

Last modified: Apr 20, 2006 0946:15 since March 29, 2006

start Siesta-school Cairo-08 Pseudopotentials SIESTA (Spanish Initi... 5:41 PM

**Warning:**

**Before use a pseudopotential for production,  
test it**

**Independent of SIESTA or any other code**

# What are the main approximations?

## Born-Oppenheimer

Decouple the movement of the electrons and the nuclei.

## Density Functional Theory (talk by Notker Roesch)

Treatment of the electron – electron interactions.

## Pseudopotentials

Treatment of the (nuclei + core) – valence.

## Basis set

To expand the eigenstates of the hamiltonian.

## Numerical evaluation of matrix elements

Efficient and self-consistent computations of  $H$  and  $S$ .

## Solve the secular equation

## Supercells

To deal with periodic systems

# Most important reference followed in this lecture

phys. stat. sol. (b) **215**, 809 (1999)

Subject classification: 71.15.Mb; 71.15.Fv; 71.24.+q; S1.3; S5; S5.11

## **Linear-Scaling ab-initio Calculations for Large and Complex Systems**

E. ARTACHO<sup>1</sup>) (a), D. SÁNCHEZ-PORTAL (b), P. ORDEJÓN (c), A. GARCÍA (d),  
and J. M. SOLER (e)

PHYSICAL REVIEW B, VOLUME 64, 235111

### **Numerical atomic orbitals for linear-scaling calculations**

Javier Junquera,<sup>1</sup> Óscar Paz,<sup>1</sup> Daniel Sánchez-Portal,<sup>2,3</sup> and Emilio Artacho<sup>4</sup>

PHYSICAL REVIEW B **66**, 205101 (2002)

### **Systematic generation of finite-range atomic basis sets for linear-scaling calculations**

Eduardo Anglada,<sup>1,2</sup> José M. Soler,<sup>1</sup> Javier Junquera,<sup>3</sup> and Emilio Artacho<sup>4</sup>

# the many body problem reduced to a problem of independent particles

One particle Kohn-Sham equation

$$\left[ -\frac{1}{2} \nabla^2 + V_{eff}^{\sigma}(\vec{r}) \right] \psi_i^{\sigma}(\vec{r}) = \varepsilon_i^{\sigma} \psi_i^{\sigma}(\vec{r})$$

$$V_{eff}^{\sigma}(\vec{r}) = V_{ext}(\vec{r}) + V_{Hartree}[n] + V_{xc}^{\sigma}[n^{\uparrow}, n^{\downarrow}]$$

**Goal:** solve the equation, that is, **find**

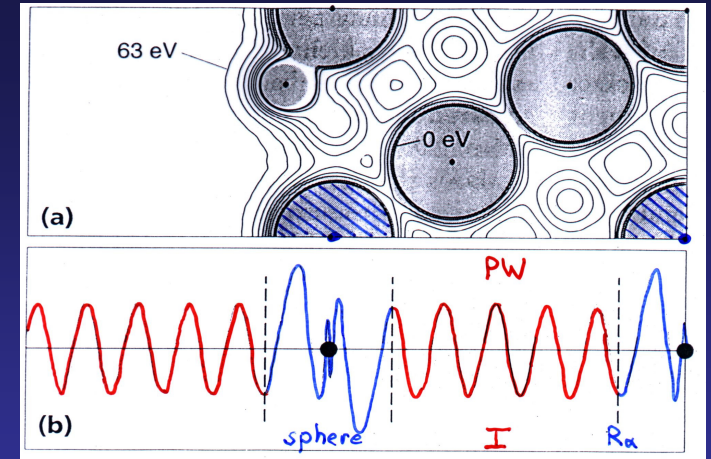
- the **eigenvectors**
- the **eigenvalues**

**Solution:** expand the eigenvectors in terms of functions of known properties (**basis**)

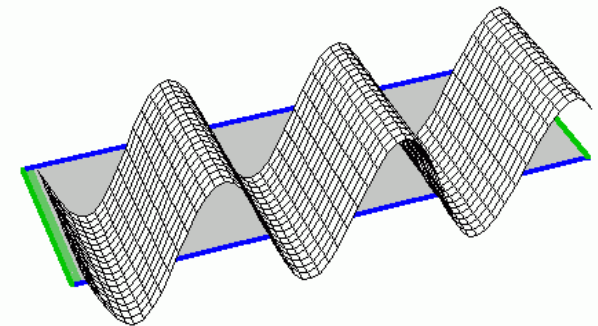
$$\psi_i(\vec{r}) = \sum_{\alpha} c_{i\alpha} \underbrace{f_{\alpha}(\vec{r})}_{\text{basis functions}}$$

# Three main families of methods depending on the basis sets

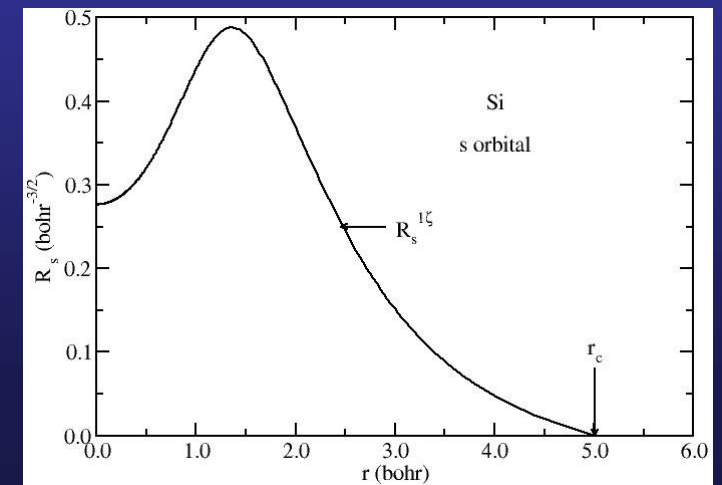
Atomic sphere methods



Plane wave and grids

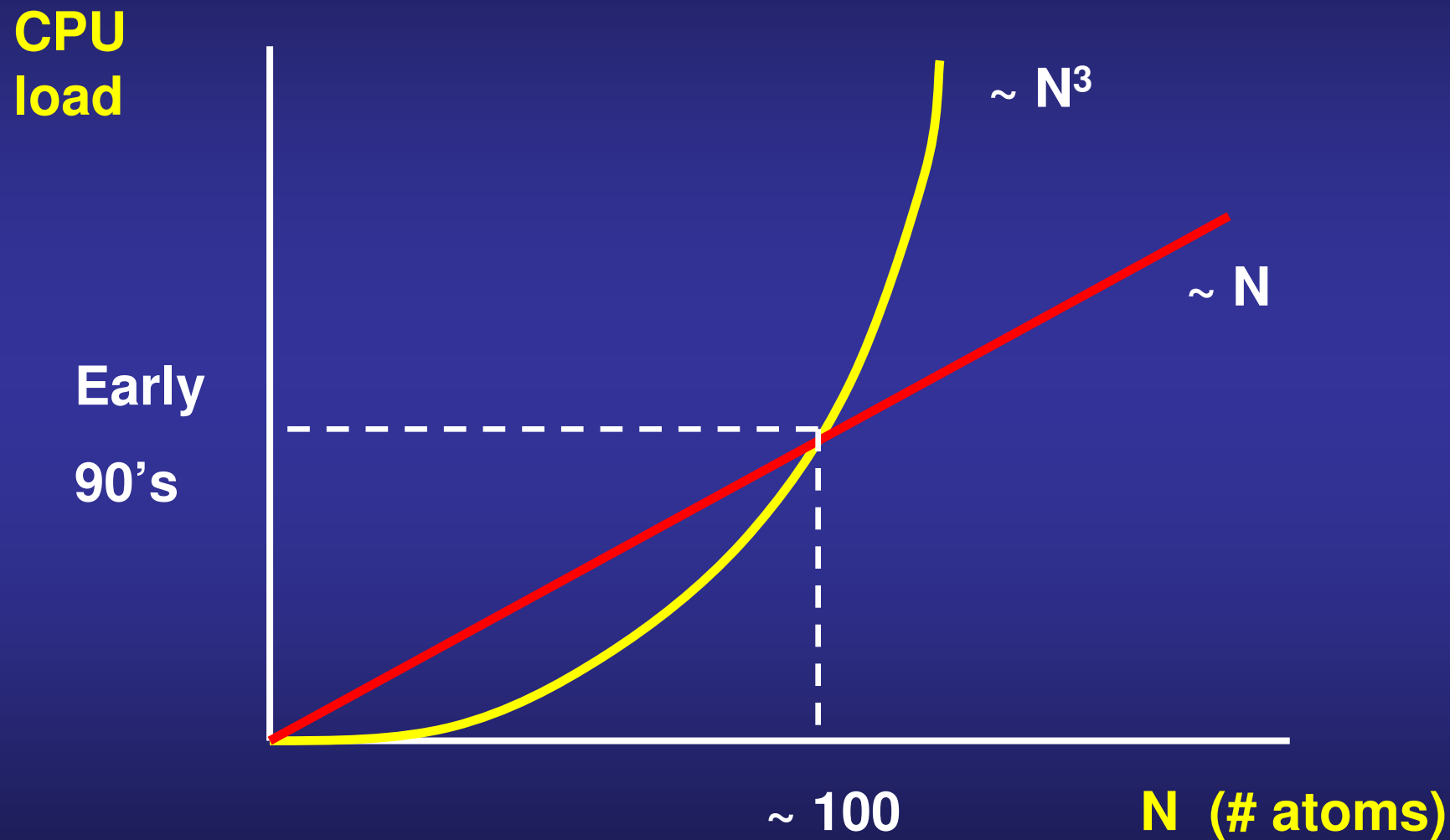


Localized basis sets





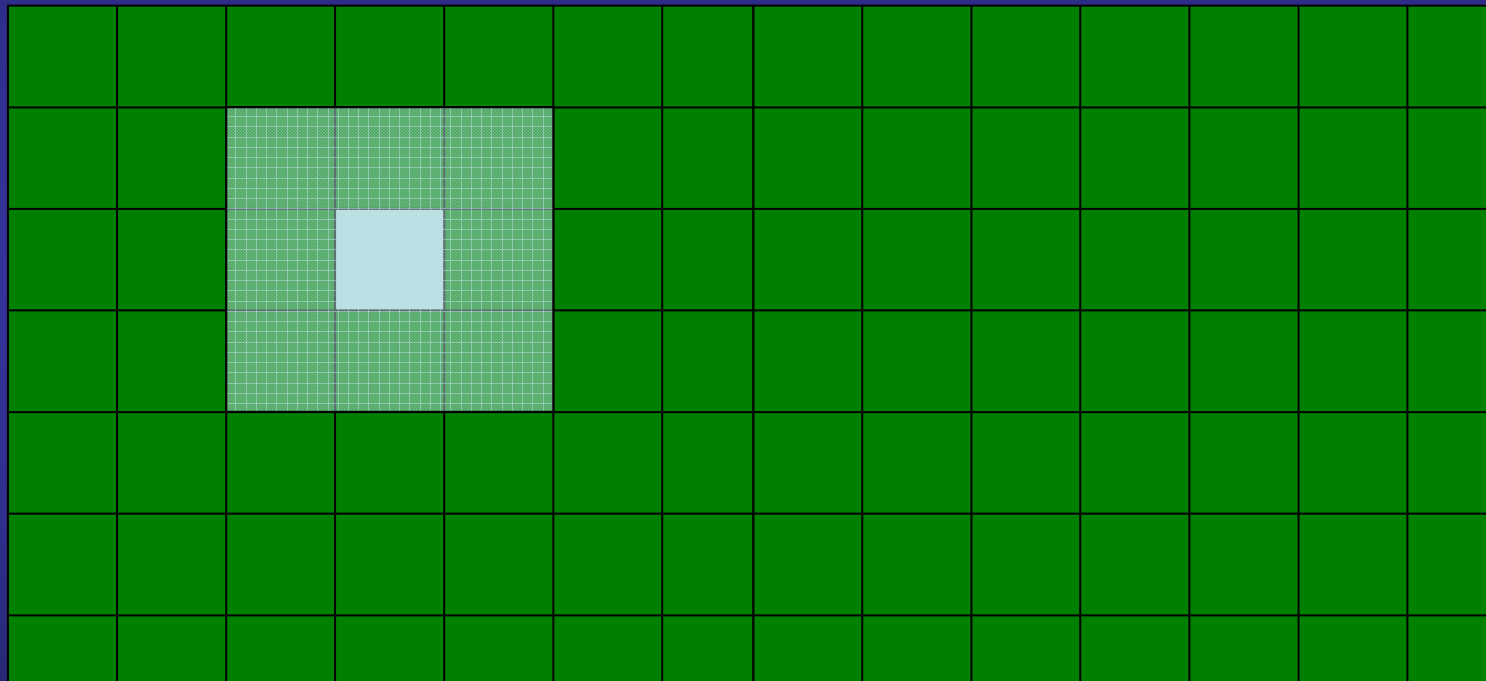
# Order-N methods: The computational load scales linearly with the system size



G. Galli and M. Parrinello, Phys. Rev Lett. 69, 3547 (1992)

# Locality is the key point to achieve linear scaling

Large system

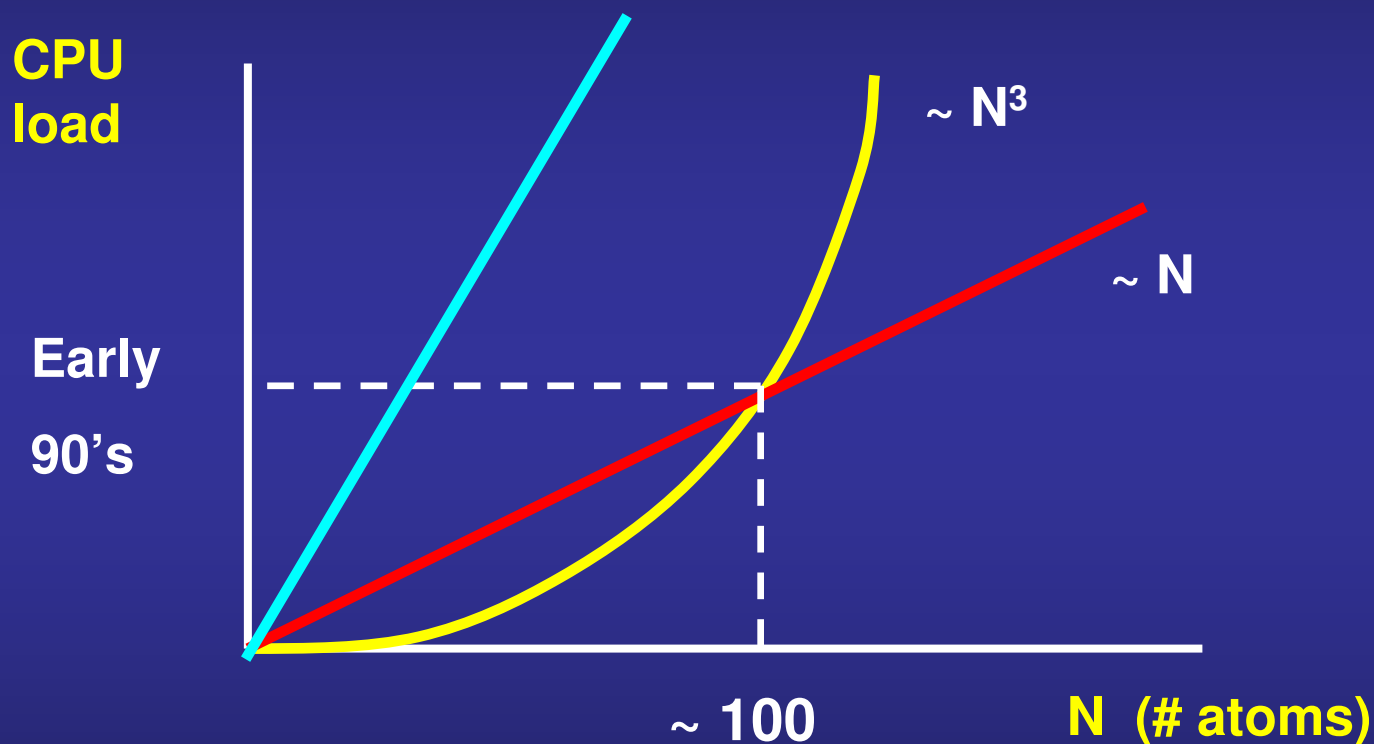


"Divide and Conquer"

W. Yang, Phys. Rev. Lett. 66, 1438 (1992)

# Efficient basis set for linear scaling calculations: localized, few and confined

**Locality**  $\Rightarrow$  Basis set of **localized functions**



Regarding **efficiency**, the important aspects are:

- **NUMBER** of basis functions per atom
- **RANGE** of localization of these functions

# Atomic orbitals: advantages and pitfalls

$$\phi_{Ilnn}(\vec{r}) = R_{Iln}(|\vec{r}_I|) Y_{lm}(\hat{r}_I)$$

## ADVANTAGES

- Very efficient (number of basis functions needed is usually very small).
- Large reduction of CPU time and memory
- Straightforward physical interpretation (population analysis, projected density of states,...)
- They can achieve very high accuracies...

## DISADVANTAGES

- ...Lack of systematic for convergence (not unique way of enlarge the basis set)
- Human and computational effort searching for a good basis set before facing a realistic project.
- Depend on the atomic position (Pulay terms).

# Atomic orbitals: a radial function times an spherical harmonic

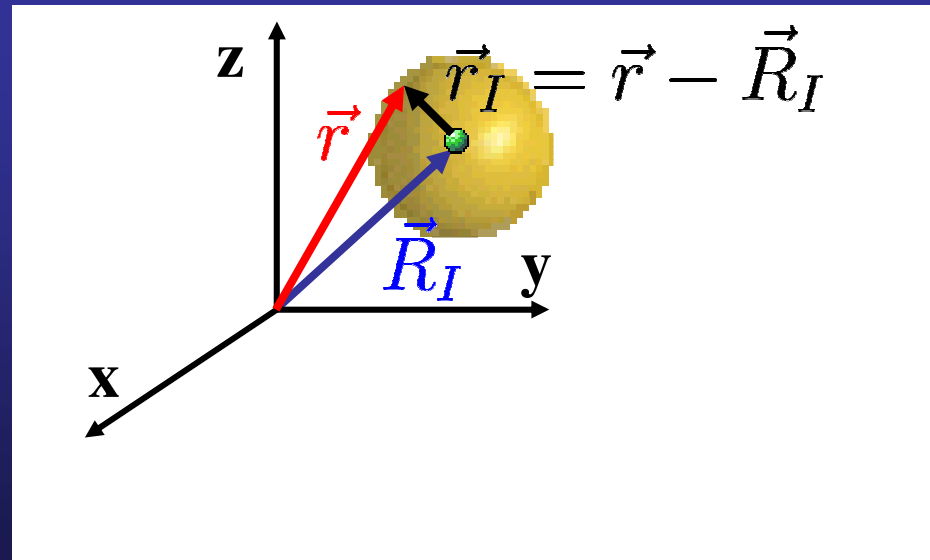
$$\phi_{Ilmn}(\vec{r}) = R_{Ilm}(|\vec{r}_I|) Y_{lm}(\hat{r}_I)$$

Index of an atom  $\nearrow$

Possibility of multiple orbitals with the same  $l, m$   $\nearrow$

Angular momentum  $\uparrow$

$$\hat{r}_I = \frac{\vec{r}_I}{|\vec{r}_I|}$$

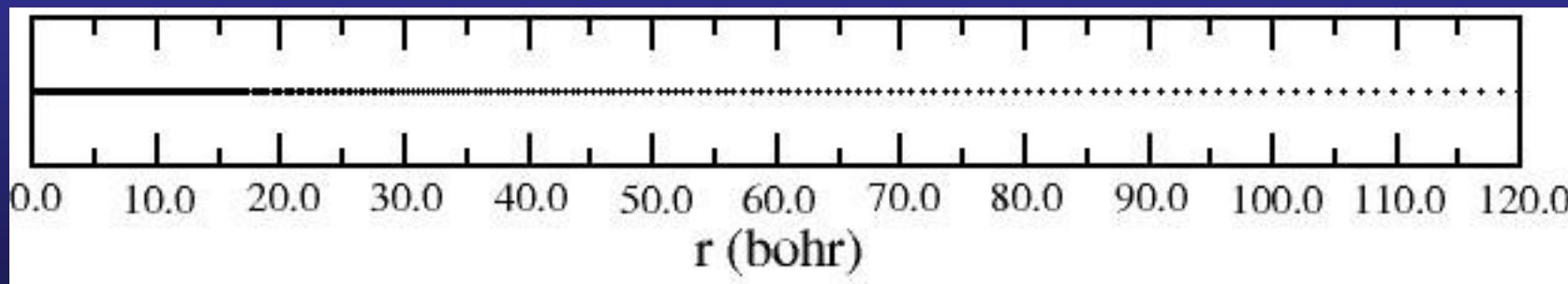


# Numerical atomic orbitals

**Numerical solution** of the Kohn-Sham Hamiltonian for the **isolated pseudoatom** with the **same approximations** (xc,pseudos) as for the condensed system

$$\left( -\frac{1}{2r} \frac{d^2}{dr^2} r + \frac{l(l+1)}{2r^2} + V_l(r) \right) R_l(r) = \varepsilon_l R_l(r)$$

This equation is solved in a logarithmic grid using the Numerov method



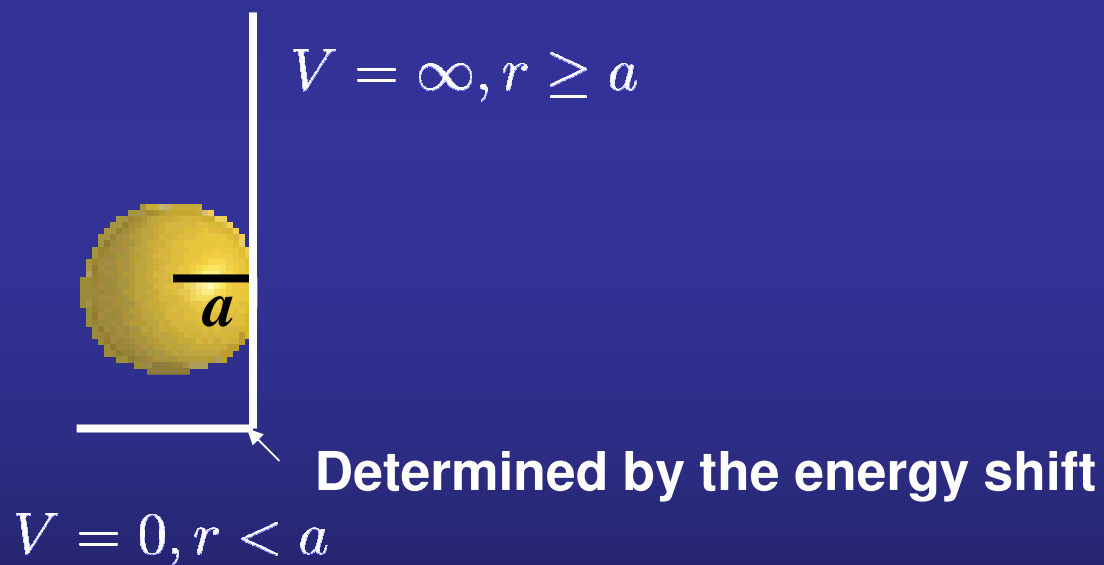
Dense close at the origin where  
atomic quantities oscillates wildly

Light far away from the origin where  
atomic quantities change smoothly

# In SIESTA: strictly localized numerical atomic orbitals

The Schrödinger equation for the isolated atom is solved within a confinement potential that forces the orbital to be extrivtly zero beyond a given  $r_c$ .

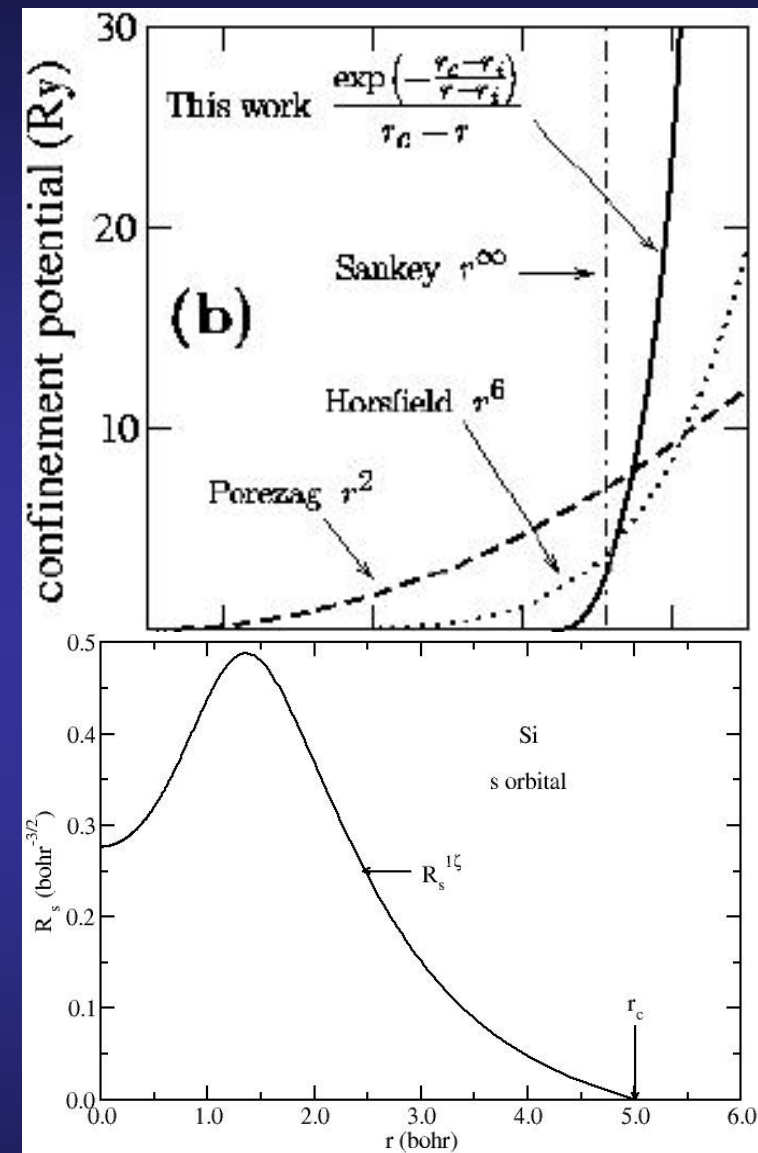
## The default in SIESTA



## Fireball

O. F. Sankey and D. J. Niklewski, Phys. Rev. B 40, 3979 (89)

Empirically, it works very nice



# Atomic orbitals:

## Main features that characterize the basis

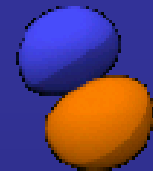
$$\phi_{Ilmn}(\vec{r}) = R_{Il}(r) Y_{lm}(\hat{r})$$

Spherical harmonics:  
well defined (fixed) objects

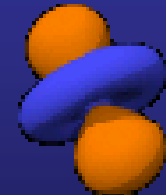
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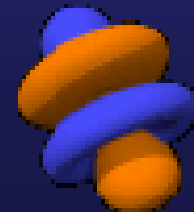
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d



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# We use real spherical harmonics for computational efficiency

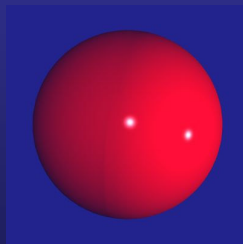
$$Y_{lm}(\theta, \varphi) = C_{lm} P_l^m(\cos\theta) \begin{cases} \sin(m\varphi) & \text{if } m < 0 \\ \cos(m\varphi) & \text{if } m \geq 0 \end{cases}$$

Normalization factors

Associated Legendre polynomials

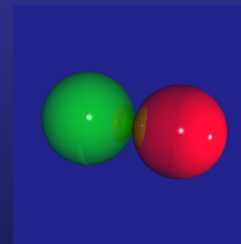
$l = 0$

$m = 0$

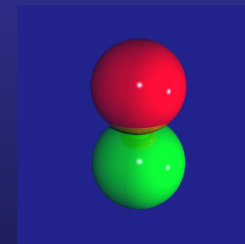


$l = 1$

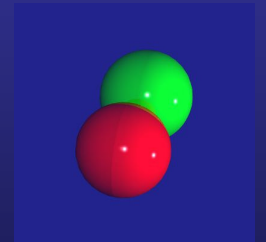
$m = -1$



$m = 0$



$m = +1$



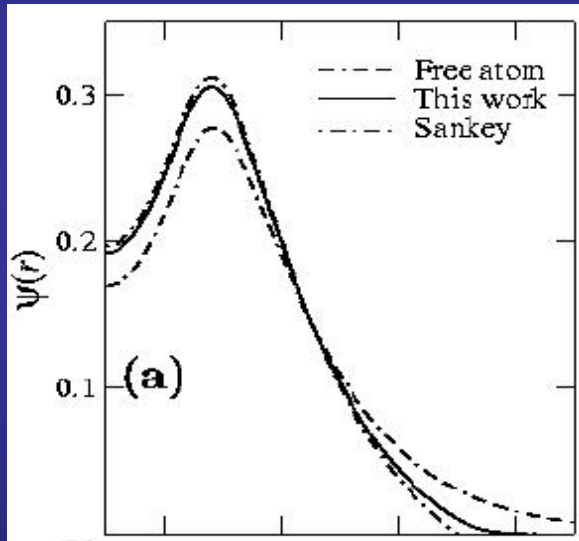
Pictures courtesy of Victor Luaña

# Atomic orbitals:

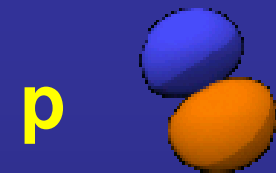
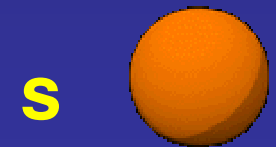
## Main features that characterize the basis

$$\phi_{Ilmn}(\vec{r}) = R_{Ilm}(|\vec{r}_I|) Y_{lm}(\hat{r}_I)$$

Radial part:  
degree of freedom to play with



Spherical harmonics:  
well defined (fixed) objects



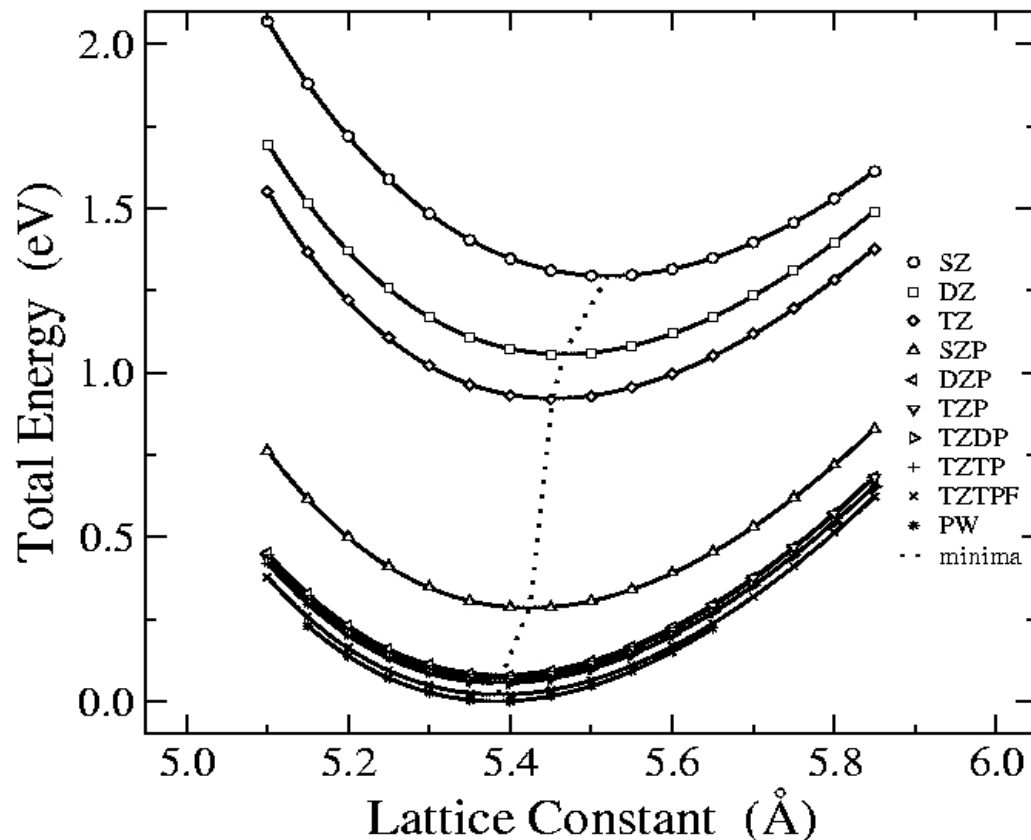
**Size:** Number of atomic orbitals per atom

**Range:** Spatial extension of the orbitals

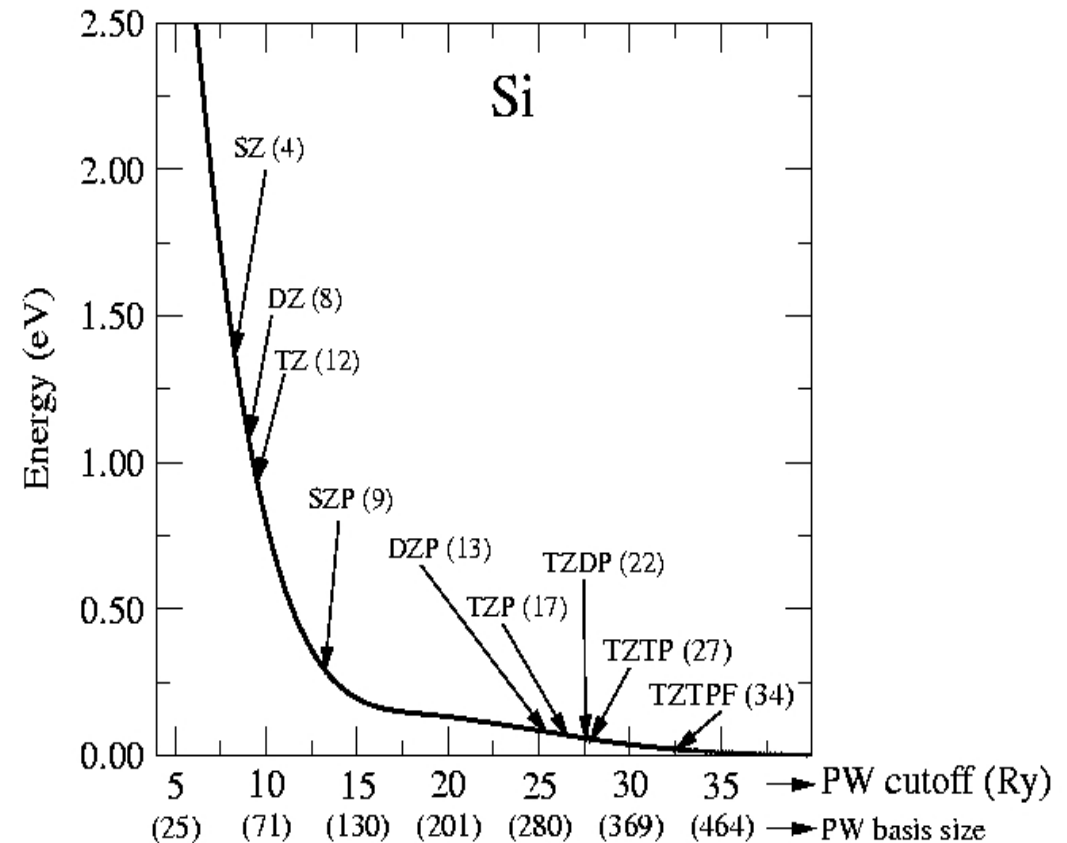
**Shape:** of the radial part

# Convergence as a function of the size of the basis set: Bulk Si

## Cohesion curves



## PW and NAO convergence



Atomic orbitals show nice convergence with respect the size

Polarization orbitals very important for convergence (more than multiple- $\zeta$ )

Double- $\zeta$  plus polarization equivalent to a PW basis set of 26 Ry

# Convergence as a function of the size of the basis set: Bulk Si

	SZ	DZ	TZ	SZP	DZP	TZP	TZDP	PW	APW	Exp
a (Å)	5.52	5.46	5.45	5.42	5.39	5.39	5.39	5.38	5.41	5.43
B (GPa)	89	96	98	98	97	97	96	96	96	98.8
E <sub>c</sub> (eV)	4.72	4.84	4.91	5.23	5.33	5.34	5.34		5.28	4.63

**A DZP basis set introduces the same deviations as the ones due to the DFT or the pseudopotential approaches**

SZ = single- $\zeta$

DZ= doble-  $\zeta$

TZ=triple-  $\zeta$

P=Polarized

DP=Doble-polarized

PW: Converged Plane Waves (50 Ry)

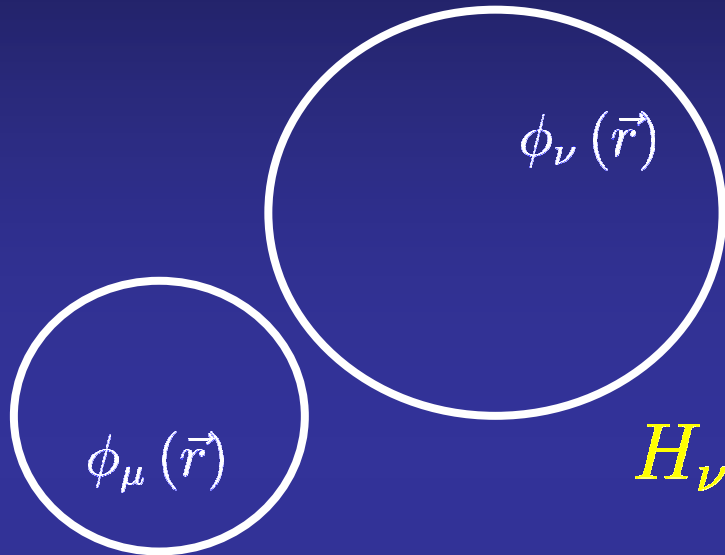
APW: Augmented Plane Waves

# Convergence as a function of the size of the basis set

		Exp	LAPW	Other PW	PW	DZP
Au	$a$	4.08 <sup>a</sup>	4.05 <sup>b</sup>	4.07 <sup>c</sup>	4.05	4.07
	$B$	173 <sup>a</sup>	198 <sup>b</sup>	190 <sup>c</sup>	191	188
	$E_c$	3.81 <sup>a</sup>	-	-	4.19	4.03
MgO	$a$	4.21 <sup>d</sup>	4.26 <sup>e</sup>	-	4.10	4.11
	$B$	152 <sup>d</sup>	147 <sup>e</sup>	-	168	167
	$E_c$	10.30 <sup>d</sup>	10.40 <sup>e</sup>	-	11.90	11.87
C	$a$	3.57 <sup>a</sup>	3.54 <sup>f</sup>	3.54 <sup>g</sup>	3.53	3.54
	$B$	442 <sup>a</sup>	470 <sup>f</sup>	436 <sup>g</sup>	466	453
	$E_c$	7.37 <sup>a</sup>	10.13 <sup>f</sup>	8.96 <sup>g</sup>	8.90	8.81
Si	$a$	5.43 <sup>a</sup>	5.41 <sup>h</sup>	5.38 <sup>g</sup>	5.38	5.40
	$B$	99 <sup>a</sup>	96 <sup>h</sup>	94 <sup>g</sup>	96	97
	$E_c$	4.63 <sup>a</sup>	5.28 <sup>h</sup>	5.34 <sup>g</sup>	5.37	5.31
Na	$a$	4.23 <sup>a</sup>	4.05 <sup>i</sup>	3.98 <sup>g</sup>	3.95	3.98
	$B$	6.9 <sup>a</sup>	9.2 <sup>i</sup>	8.7 <sup>g</sup>	8.8	9.2
	$E_c$	1.11 <sup>a</sup>	1.44 <sup>j</sup>	1.28 <sup>g</sup>	1.22	1.22
Cu	$a$	3.60 <sup>a</sup>	3.52 <sup>b</sup>	3.56 <sup>g</sup>	-	3.57
	$B$	138 <sup>a</sup>	192 <sup>b</sup>	172 <sup>g</sup>	-	165
	$E_c$	3.50 <sup>a</sup>	4.29 <sup>k</sup>	4.24 <sup>g</sup>	-	4.37
Pb	$a$	4.95 <sup>a</sup>	-	4.88	-	4.88
	$B$	43 <sup>a</sup>	-	54	-	64
	$E_c$	2.04 <sup>a</sup>	-	3.77	-	3.51

# Range: the spatial extension of the atomic orbitals

Order(N) methods  $\Rightarrow$  locality, that is, a finite range for matrix and overlap matrices



If the two orbitals are sufficiently far away

$$S_{\nu\mu} = \langle \phi_\nu | \phi_\mu \rangle = \int d\vec{r} \phi_\nu^*(\vec{r}) \phi_\mu(\vec{r}) = 0$$

$$H_{\nu\mu} = \langle \phi_\nu | \hat{H} | \phi_\mu \rangle = \int d\vec{r} \phi_\nu^*(\vec{r}) \hat{H} \phi_\mu(\vec{r}) = 0$$

## Neglect interactions:

Below a tolerance

Beyond a given scope of neighbours

**Problem:** introduce numerical instabilities for high tolerances.

## Strictly localized atomic orbitals:

Vanishes beyond a given cutoff radius

O. Sankey and D. Niklewski, PRB 40, 3979 (89)

**Problem:** accuracy and computational efficiency depend on the range of the basis orbitals

How to define all the  $r_c$  in a balance way?

# How to control the range of the orbitals in a balanced way: the energy shift

$$\left( -\frac{1}{2r} \frac{d^2}{dr^2} r + \frac{l(l+1)}{2r^2} + V_l(r) \right) R_l(r) = (\varepsilon_l + \delta\varepsilon_l) R_l(r)$$



Energy increase  $\equiv$  Energy shift

PAO.EnergyShift (energy)

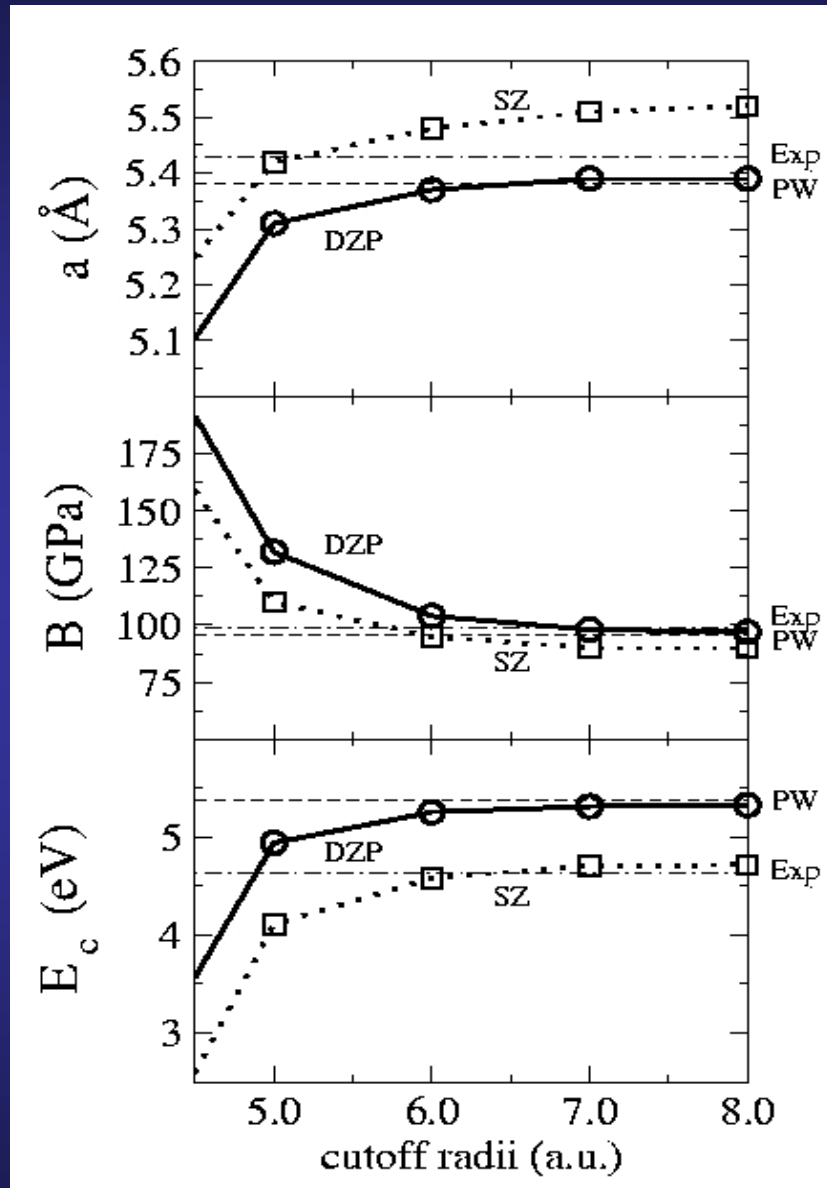
Cutoff radius,  $r_c$ , = position where each orbital has the node

A **single parameter for all** cutoff radii

The larger the Energy shift, the shorter the  $r_c$ s

Typical values: 100-200 meV

# Convergence with the range



**Bulk Si**  
**equal  $s, p$   
orbitals radii**

J. Soler *et al.*, J. Phys: Condens. Matter, 14, 2745 (2002)

More efficient



More accurate



# How to introduce the basis set in SIESTA

## Effort on defining a systematics with minimum parameters

If **nothing** is specified: **default**

Basis size:	PAO.BasisSize	DZP
Range of first-zeta:	PAO.EnergyShift	0.02 Ry
Second-zeta:	PAO.BasisType	Split
Range of second-zeta:	PAO.SplitNorm	0.15
Confinement:	Hard well	

Good basis set in terms of accuracy versus efficiency

The user has the freedom to play with these parameters

Recently: used variationally optimized basis set

# As with the pseudopotentials, there is a list of contributed basis sets


SIESTA (Spanish Initiative for Electronic Simulations with Thousands of Atoms) - Mozilla Firefox


Archivo Editar Ver Historial Marcadores Herramientas Ayuda

http://www.uam.es/departamentos/ciencias/fismateriac/siesta/

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webmaster:  
siesta.web@uam.es

## SIESTA pseudopotential database

### Ti: pseudopotentials

- [Upload a pseudopotential](#)
- [Upload a basis set for one of these pseudopotentials](#)

### Available pseudopotentials:


[Ti semicore states \(3s and 3p\) in valence](#) (Author: Javier Junquera; created on 13/10/2005)

Flavour of the pseudopotential: Troullier-Martins. Exchange and correlation functional: LDA (Ceperley-Alder). Relativistic: yes. Core corrections: no. Valence reference configuration: 3s<sup>2</sup> 3p<sup>6</sup> 3d<sup>2</sup> 4f<sup>0</sup> (ionic configuration, ionic charge +2). Cutoff radius: 3s 1.30 bohr 3p 1.30 bohr 3d 1.30 bohr 4f 2.00 bohr

### Available basis sets:

[Ti semicore states \(3s and 3p\) in valence](#) (Author: Javier Junquera; created on 13/10/2005)

Basis set optimized with Simplex for the bulk cubic cell of BaTiO<sub>3</sub>. The semicore states 3s and 3p are included in the valence. The quality of the basis is DZP for the valence (two radial functions for the 4s, and two radial functions for the 3d state, and one radial function for the 4p states), and SZ for the semicore states (one radial function for the 3s and the 3p). This amounts to 19 basis functions per Ti atom.



Generated by [psbase](#): (c) [Andrew Walkingshaw](#), 2003.

W3C HTML 4.01 W3C CSS

Terminado

start Siesta-school Cairo-08 Pseudopotentials SIESTA (Spanish Initi...

0946:15 since March 29, 2006

6:12 PM

# What are the main approximations?

## Born-Oppenheimer

Decouple the movement of the electrons and the nuclei.

## Density Functional Theory (talk by Notker Roesch)

Treatment of the electron – electron interactions.

## Pseudopotentials

Treatment of the (nuclei + core) – valence.

## Basis set

To expand the eigenstates of the hamiltonian.

## Numerical evaluation of matrix elements

Efficient and self-consistent computations of  $H$  and  $S$ .

## Solve the secular equation

## Supercells

To deal with periodic systems

# Goal: solve the one-particle Kohn-Sham Schrödinger-like equation

$$\hat{H}\psi_i(\vec{r}) = E_i\psi_i(\vec{r})$$

Expansion of the eigenvectors in a basis of localized atomic orbitals

$$\psi_i(\vec{r}) = \sum_{\mu} \phi_{\mu}(\vec{r}) c_{\mu i}$$

where the coefficients  $c_{\mu i} = \langle \tilde{\phi}_{\mu} | \psi_i \rangle$ , and  $\tilde{\phi}_{\mu}$  are the dual orbital of  $\phi_{\mu}$ :  $\langle \tilde{\phi}_{\mu} | \phi_{\nu} \rangle = \delta_{\mu\nu}$

Introducing the expansion into the Kohn-Sham equation, we arrive to the **secular equation**

$$\sum_{\mu} (H_{\nu\mu} - E_i S_{\nu\mu}) c_{\mu i} = 0$$

$$S_{\nu\mu} = \langle \phi_{\nu} | \phi_{\mu} \rangle = \int d\vec{r} \phi_{\nu}^*(\vec{r}) \phi_{\mu}(\vec{r})$$

$$H_{\nu\mu} = \langle \phi_{\nu} | \hat{H} | \phi_{\mu} \rangle = \int d\vec{r} \phi_{\nu}^*(\vec{r}) \hat{H} \phi_{\mu}(\vec{r})$$

# The one-particle Kohn-Sham hamiltonian

$$\hat{H} = \hat{T} + \sum_{\alpha} \hat{V}_{\alpha}^{PS} + V^H(\vec{r}) + V^{xc}(\vec{r})$$

Transforming the semilocal pseudopotential form into the fully nonlocal separable Kleinman-Bylander form

$$\hat{V}^{PS} = V^{local}(\vec{r}) + \hat{V}^{KB}$$

$$V^{local}(\vec{r}) \rightarrow -\frac{Z_{val}}{r} \qquad \hat{V}^{KB} = \sum_{l=0}^{l_{max}^{KB}} \sum_{m=-l}^l \sum_{n=1}^{N_l^{KB}} |\chi_{lmn}^{KB}\rangle v_{ln}^{KB} \langle \chi_{lmn}^{KB}|$$

The standard Kohn-Sham one-electron hamiltonian might be written as

$$\hat{T} + \sum_{\alpha} V_{\alpha}^{local}(\vec{r}) + \sum_{\alpha} V_{\alpha}^{KB} + V^H(\vec{r}) + V^{xc}(\vec{r})$$

Kinetic energy operator

$$\hat{T} = -\frac{1}{2}\nabla^2$$

Hartree potential

$$V^H(\vec{r}) = \int d\vec{r}' \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|}$$

Exchange-correlation potential

(Assume LDA approach)

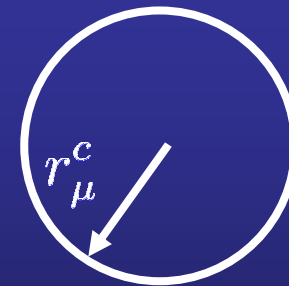
$$V^{xc}(\vec{r}) = V^{xc}[\rho(\vec{r})]$$

**Electronic charge density =**  
**sum of spherical atomic densities +**  
**deformation charge density (bonding)**

$$\rho(\vec{r}) = \rho^{atom}(\vec{r}) + \delta\rho(\vec{r})$$

$$\rho^{atom}(\vec{r}) = \sum_I \rho_I^{atom}(\vec{r})$$

↑  
 Populate basis function with  
 appropriate valence atomic charges



$\phi_\mu(\vec{r})$

$\rho_I^{atom}$  exactly vanishes beyond  $r_I^c = \max_l r_{Il}^c$

# The local part is screened by the potential generated by an atomic electron density

$$V^H(\vec{r}) = \int d\vec{r}' \frac{\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} = \int d\vec{r}' \frac{\rho^{atom}(\vec{r}')}{|\vec{r} - \vec{r}'|} + \int d\vec{r}' \frac{\delta\rho(\vec{r}')}{|\vec{r} - \vec{r}'|} \equiv V^{atom}(\vec{r}) + \delta V^H(\vec{r})$$

Neutral atom potential

$$V^{local}(\vec{r}) \rightarrow -\frac{Z_{val}}{r}$$

$$V_I^{NA}(\vec{r}) \equiv V_I^{local}(\vec{r}) + V_I^{atom}(\vec{r})$$

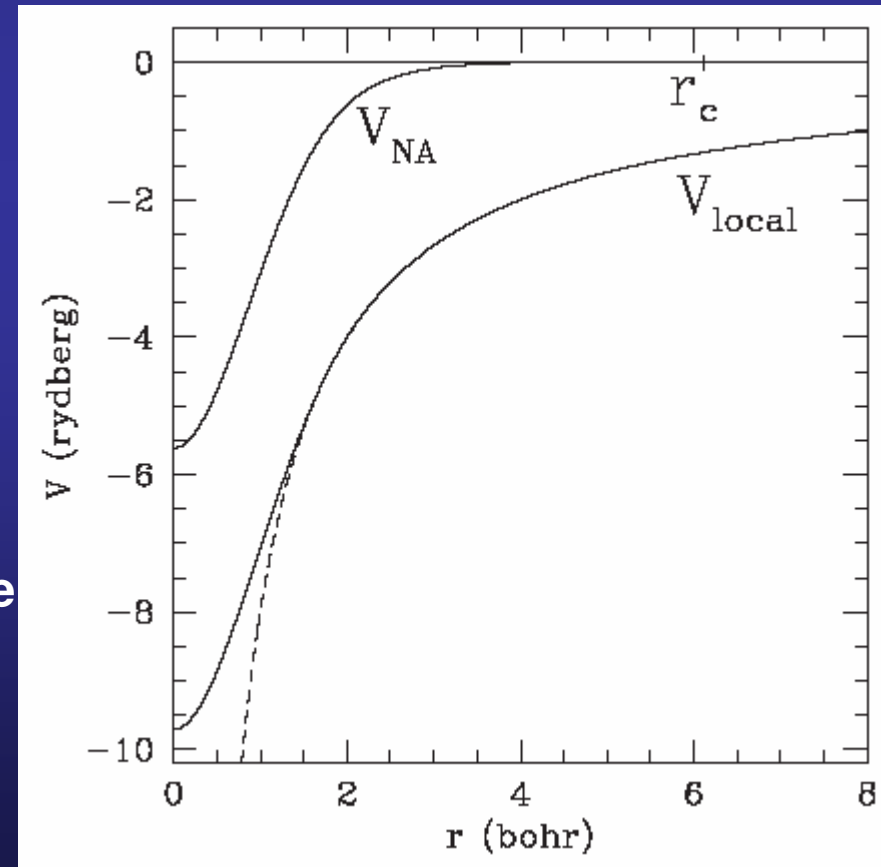


$$V_I^{NA}(\vec{r}) = 0$$

Potential outside the sphere vanishes

(Gauss theorem  $\Rightarrow$   
generated by the total  
charge inside the sphere  
= 0 if neutral atom)

Vanishes exactly at  $r_c$



# The hamiltonian computed in SIESTA, combination of two and three center matrix elements

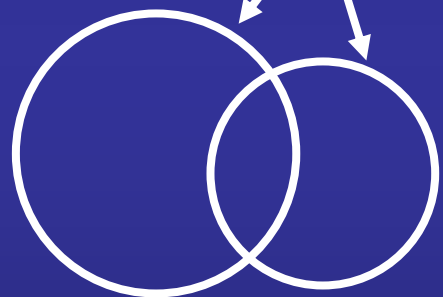
Two center integrals

Three center integrals

$$\hat{H} = \hat{T} + \hat{V}^{NL} + V^{NA}(\vec{r}) + \delta V^H(\vec{r}) + V^{xc}(\vec{r})$$

$$\langle \phi_\nu | \nabla^2 | \phi_\mu \rangle$$

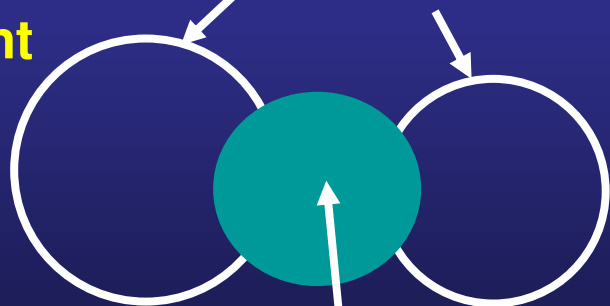
Basis orbitals



Non self-consistent

$$\langle \phi_\nu | \chi_{lmn}^{KB} \rangle \langle \chi_{lmn}^{KB} | \phi_\mu \rangle$$

Basis orbitals



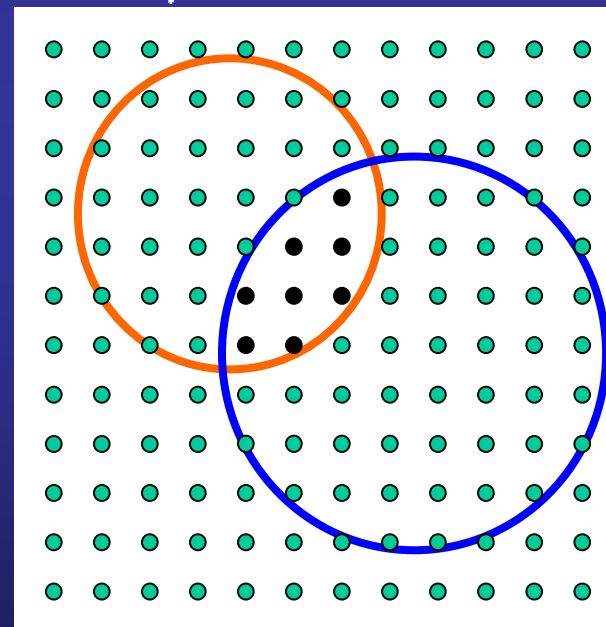
KB pseudopotential projector

Computed in reciprocal space and tabulated

$$\langle \phi_\nu | V(\vec{r}) | \phi_\mu \rangle$$

$$\phi_\mu(\vec{r})$$

Self-consistent

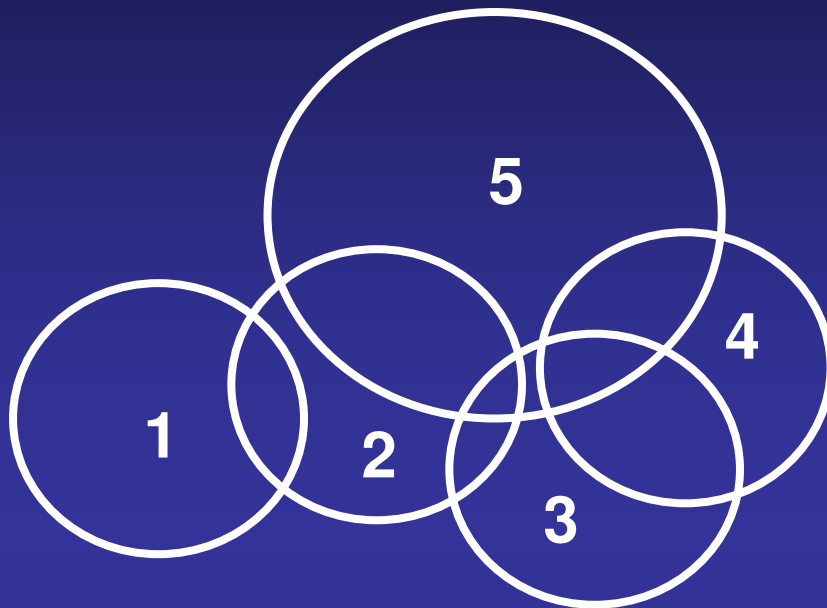
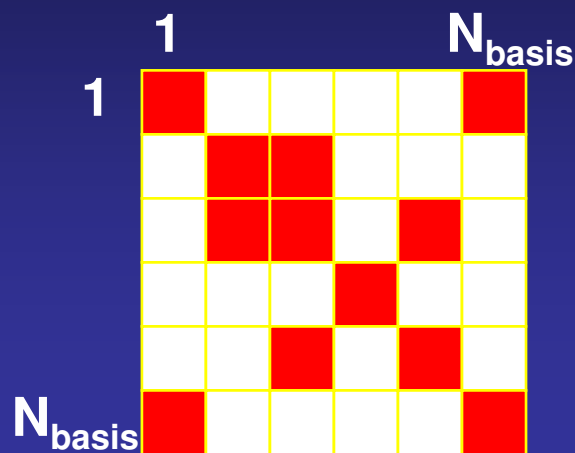


$$\phi_\nu(\vec{r})$$

Three-dimensional real space grid



# Order-N methods rely heavily on the sparsity of the Hamiltonian and overlap matrices



1 with 1 and 2

2 with 1,2,3, and 5

3 with 2,3,4, and 5

4 with 3,4 and 5

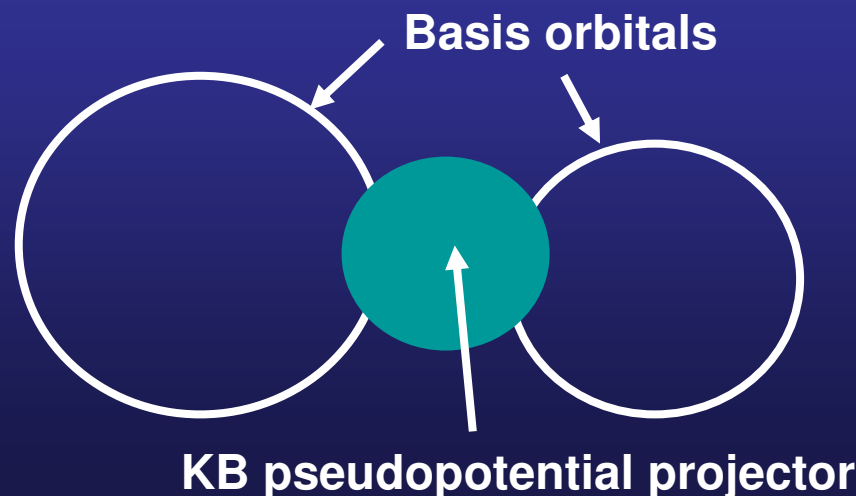
5 with 2,3,4, and 5

Sparse  $\equiv$  many entrances of the matrix are zero

$S_{\mu\nu}$  and  $H_{\mu\nu}$  are sparse

$\rho_{\mu\nu}$  is not strictly sparse but only a sparse subset is needed

## Non-overlap interactions



# Two center integrals are calculated in Fourier space

Two center integrals (i. e. the overlap) have a form like

$$S_{12}(\vec{R}) \equiv \langle \psi_1 | \psi_2 \rangle = \int_{all \ space} d\vec{r} \ \psi_1^* (\vec{r}) \ \psi_2(\vec{r} + \vec{R})$$

$\psi_1, \psi_2$  might be atomic orbitals, KB projectors or other functions centered on atoms

$S_{12}(\vec{R})$  can be seen as a convolution: in 1D  $f * g \equiv \frac{1}{\sqrt{2\pi}} \int_{-\infty}^{+\infty} g(y) f(x - y) dy$

Arfken, Mathematical Methods for Physicist, Ch 15.5

Take the Fourier transform of one of the functions

$$\psi(\vec{k}) = \frac{1}{(2\pi)^{3/2}} \int \psi(\vec{r}) e^{-i\vec{k} \cdot \vec{r}} d\vec{r}$$

The Fourier transform of a convolution in real space is a product in reciprocal space

$$S_{12}(\vec{R}) = \int d\vec{k} \ \psi_1^*(\vec{k}) \psi_2(\vec{k}) e^{-i\vec{k} \cdot \vec{R}}$$

## Two center integrals are calculated in Fourier space

$$S_{12}(\vec{R}) = \int d\vec{k} \, \psi_1^*(\vec{k}) \psi_2(\vec{k}) e^{-i\vec{k} \cdot \vec{R}}$$

For each pair of functions they are calculated and stored in a fine radial grid (2500 Ry) as a function of  $R_i$ , up to the maximum distance  $R_{max} = r_1^c + r_2^c$

The value at arbitrary distances can be obtained by accurate cubic spline interpolation (once obtained, the fine grid does not suppose a penalty in execution time, since interpolation effort is independent of the number of grid points).

# The density matrix, a basic ingredient of SIESTA

Expansion of the eigenvectors in a basis of localized atomic orbitals

$$\psi_i(\vec{r}) = \sum_{\mu} \phi_{\mu}(\vec{r}) c_{\mu i}$$

where the coefficients  $c_{\mu i} = \langle \tilde{\phi}_{\mu} | \psi_i \rangle$ , and  $\tilde{\phi}_{\mu}$  are the dual orbital of  $\phi_{\mu}$ :  $\langle \tilde{\phi}_{\mu} | \phi_{\nu} \rangle = \delta_{\mu\nu}$

The electron density is given by

$$\rho(\vec{r}) = \sum_i n_i |\psi_i(\vec{r})|^2$$

Occupation of state  $\psi_i$

Inserting the expansion into the definition of the density

$$\rho(\vec{r}) = \sum_{\mu\nu} \rho_{\mu\nu} \phi_{\nu}^*(\vec{r}) \phi_{\mu}(\vec{r})$$

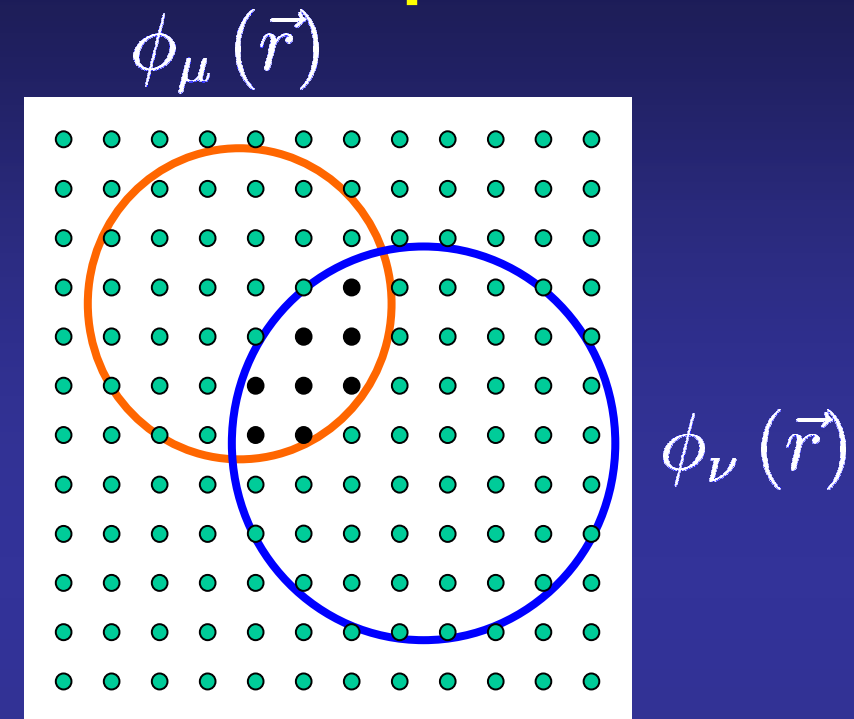
where, with  $c_{i\nu} \equiv c_{\nu i}^*$ , the **density matrix** is defined

$$\rho_{\mu\nu} = \sum_i c_{\mu i} n_i c_{i\nu}$$

**Control convergence SCF**  
**Restart calculations**

# Three dimensional grid to compute Hartree, exchange correlation and neutral atom potentials

$$\rho(\vec{r}) = \sum_{\mu\nu} \rho_{\mu\nu} \phi_{\nu}^*(\vec{r}) \phi_{\mu}(\vec{r})$$



Find all the atomic orbitals that do not vanish at a given grid point

(in practice, interpolate the radial part from numerical tables)

Once the density is known, we compute the potentials

**EVERYTHING  $O(N)$**

$$\rho(\vec{r}) \rightarrow V^{xc}(\vec{r})$$

$$\delta\rho(\vec{r}) = \rho(\vec{r}) - \rho_{atoms}(\vec{r})$$

$$\delta\rho(\vec{r}) \xrightarrow{FFT} \delta V^H(\vec{r})$$

# The Poisson equation is solved in the real space grid by FFTs

$$\nabla^2 V^H(\vec{r}) = -4\pi\rho(\vec{r})$$

Since the unit cell is periodic (naturally or artificially), we can expand the density in a Fourier series

$$\rho(\vec{r}) = \sum_{\vec{G}} \rho(\vec{G}) e^{i\vec{G}\cdot\vec{r}} \Rightarrow V^H(\vec{r}) = \sum_{\vec{G}} V^H(\vec{G}) e^{i\vec{G}\cdot\vec{r}}$$

In reciprocal space, the differential Poisson equation is nothing else than a division

$$V^H(\vec{G}) = -4\pi \frac{\rho(\vec{G})}{|\vec{G}|^2}$$

Once the coefficients of the potential are known in reciprocal space, Fourier transform back to real space

$$\rho(\vec{r}) \xrightarrow{FFT} \rho(\vec{G}) \longrightarrow V^H(\vec{G}) \xrightarrow{IFFT} V^H(\vec{r})$$

FFT scales as  $N \log(N)$

However its cost is negligible and has no influence on the overall scaling properties.

Multigrid techniques (by Oswaldo Diéguez) coming soon

# Generalized Gradient Approximation, the derivative of the charge computed numerically

$$V_{xc}^{GGA}(\vec{r}) = \frac{\delta E^{GGA}[\rho(\vec{r}'), |\nabla \rho(\vec{r}')|]}{\delta \rho(\vec{r})}$$

$$V_{xc}^{GGA}\left(\rho(\vec{r}), |\nabla \rho(\vec{r})|, \nabla^2 \rho(\vec{r}), \nabla \rho(\vec{r}) \cdot \nabla |\nabla \rho(\vec{r})|\right)$$

Density gradient need not be provided, since they are  
calculated numerically using the density at the grid points

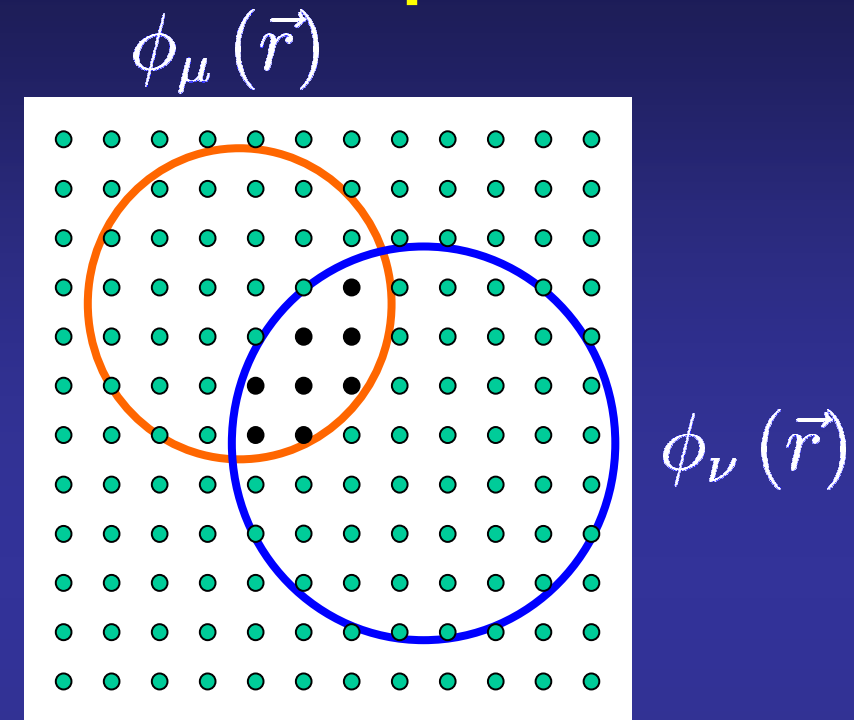
$$\frac{\partial \rho}{\partial x} = \frac{\rho_{i+1} - \rho_{i-1}}{x_{i+1} - x_{i-1}} \Rightarrow E_{xc}^{GGA}(\rho_1, \rho_2, \dots)$$

**A finer grid is required for GGA**

$$V_{xc}^{GGA}(\vec{r}_i) \equiv \frac{\partial E_{xc}^{GGA}}{\partial \rho_i}$$

# Three dimensional grid to compute Hartree, exchange correlation and neutral atom potentials

$$\rho(\vec{r}) = \sum_{\mu\nu} \rho_{\mu\nu} \phi_{\nu}^*(\vec{r}) \phi_{\mu}(\vec{r})$$



Finally, we add together all the grid contributions and perform the integral

$$V(\vec{r}) = V^{NA}(\vec{r}) + \delta V^H(\vec{r}) + V^{xc}(\vec{r})$$

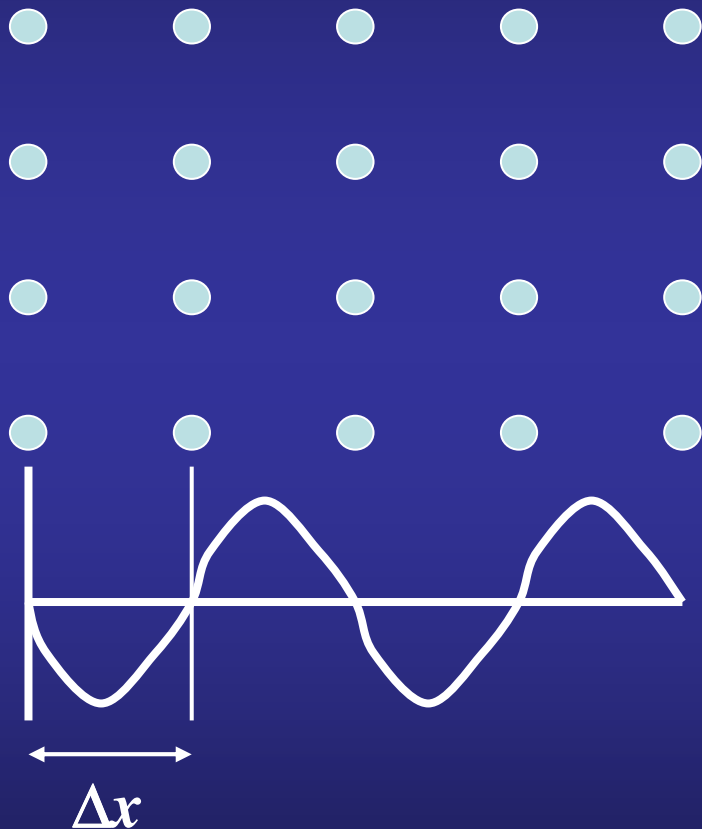
$$\int d\vec{r} \phi_{\nu}^*(\vec{r}) V(\vec{r}) \phi_{\nu}(\vec{r}) \approx \sum_i \phi_{\nu}^*(\vec{r}) V(\vec{r}) \phi_{\nu}(\vec{r}) \underset{\uparrow}{\Delta \vec{r}}$$

Volume per grid point



# Fineness of the grid controlled by a single parameter, the “MeshCutoff”

$E_{\text{cut}}$  : maximum kinetic energy of the plane waves that can be represented in the grid without aliasing



$$\Delta x \implies k_c = \frac{\pi}{\Delta x} \implies E_c = \frac{\hbar^2 k_c^2}{2m_e}$$

In the **grid**, we represent the **density**  $\Rightarrow$  grid cutoff **not directly comparable**

with the **plane wave cutoff** to represent wave functions

(Strictly speaking, the density requires a value four times larger)

# What are the main approximations?

## Born-Oppenheimer

Decouple the movement of the electrons and the nuclei.

## Density Functional Theory (talk by Notker Roesch)

Treatment of the electron – electron interactions.

## Pseudopotentials

Treatment of the (nuclei + core) – valence.

## Basis set

To expand the eigenstates of the hamiltonian.

## Numerical evaluation of matrix elements

Efficient and self-consistent computations of  $H$  and  $S$ .

## Solve the secular equation

## Supercells

To deal with periodic systems

Once the hamiltonian and the overlap matrices are build, we have to solve the Schrodinger equation

$$\begin{pmatrix} H \end{pmatrix} \begin{pmatrix} C \end{pmatrix} = E_{n\vec{k}} \begin{pmatrix} S \end{pmatrix} \begin{pmatrix} C \end{pmatrix}$$

Order-N

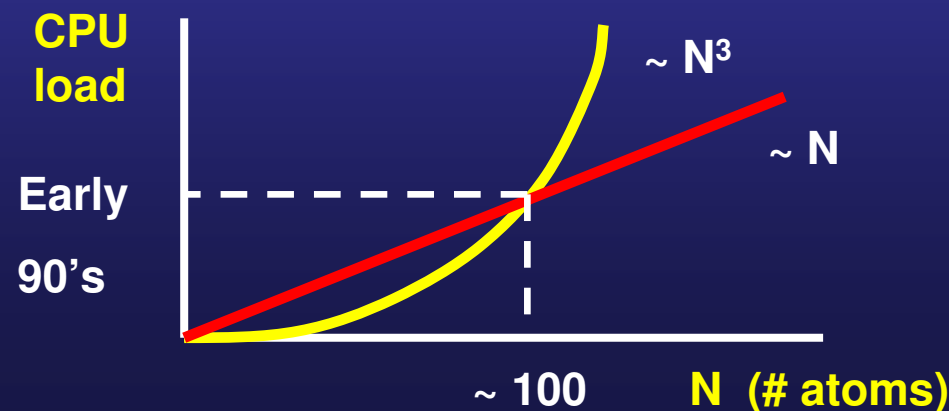
Minimization of an energy functional

Not valid for metals or “dirty” gap systems

Order-N<sup>3</sup>

Standard diagonalization techniques

Both eigenvectors and eigenvalues available



If diagonalization, the generalized eigenvalue problem is solved using standard mathematical libraries

$$\begin{pmatrix} H \\ N \times N \end{pmatrix} \begin{pmatrix} C \\ N \times 1 \end{pmatrix} = E_{n\vec{k}} \begin{pmatrix} S \\ N \times N \end{pmatrix} \begin{pmatrix} C \\ N \times 1 \end{pmatrix}$$

**Serial:**

BLAS

LAPACK

**Parallel:**

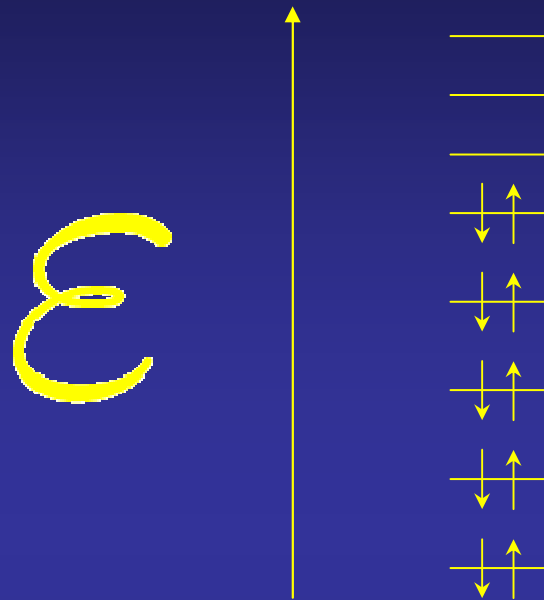
BLACS

SCALAPACK

Freely available in <http://www.netlib.org>

Most machine vendors have their own implementations available for their own platforms (acml, mkl,...).

# The one-particle eigenstates are filled following the “Aufbau” principle: from lower to higher energies



$$n^{\sigma}(\vec{r}) = \sum_i f_i^{\sigma} |\psi_i^{\sigma}(\vec{r})|^2$$

Occupation numbers

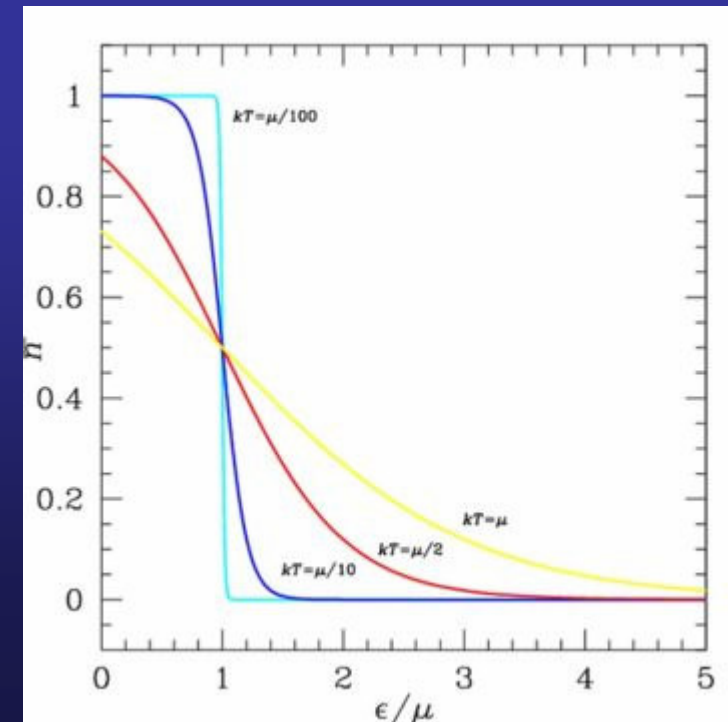
The ground state has one (or two if spin independent) in each of the orbitals with the lowest eigenvalues

A smearing of the electronic occupation might be done:

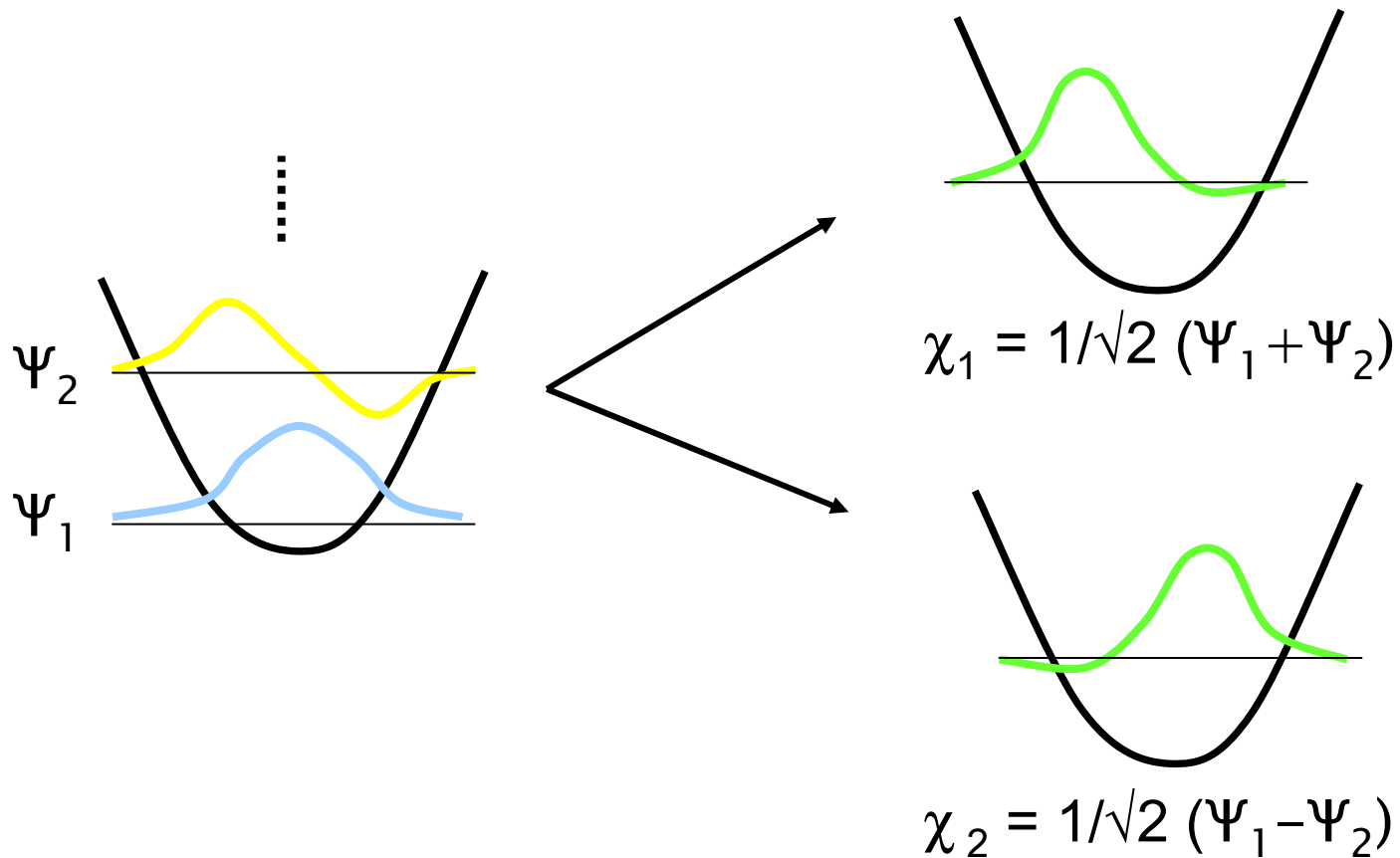
Fermi-Dirac (OccupationFunction FD)

ElectronicTemperature

Methfessel Paxton (OccupationFunction MP)



# Locality of Wave Functions



$$|\chi^{occ}\rangle = \mathbf{U}|\psi^{occ}\rangle$$

Wannier functions (crystals)

Localized Molecular Orbitals (molecules)

# Locality of Wave Functions

Energy:

$$E = \langle \psi_1 | H | \psi_1 \rangle + \langle \psi_2 | H | \psi_2 \rangle = \text{Tr}_{occ}(H)$$

Unitary Transformation:  $\{|\psi_i\rangle\} \rightarrow \{|\chi_i\rangle\}$

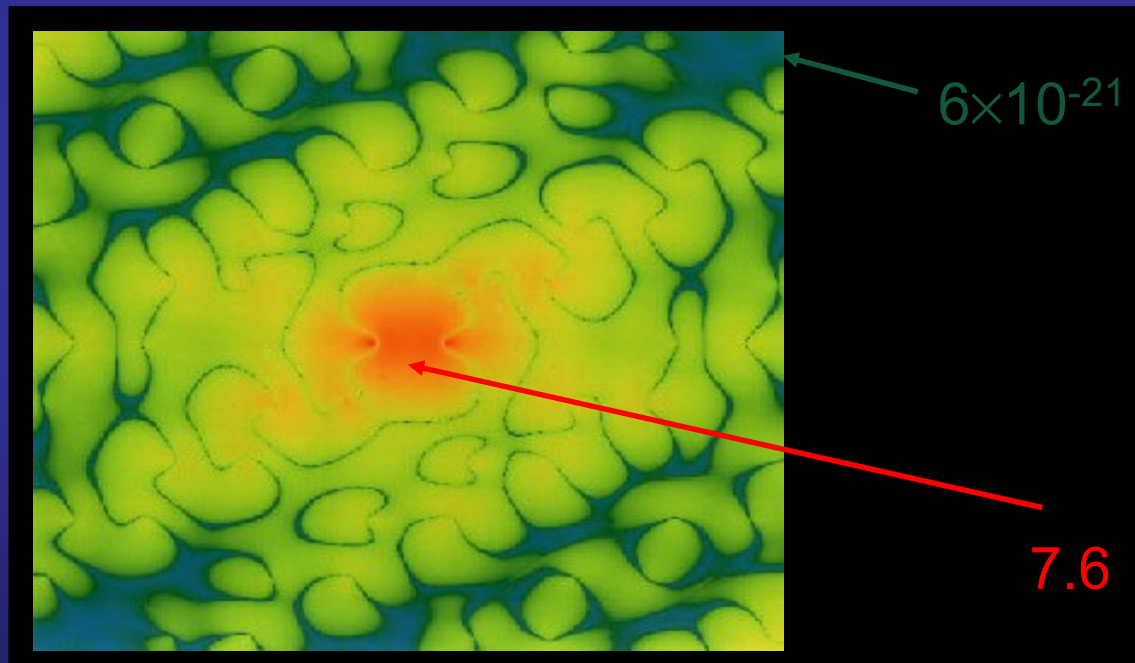
$$E = \text{Tr}_{occ}(H) = \langle \chi_1 | H | \chi_1 \rangle + \langle \chi_2 | H | \chi_2 \rangle$$

We do NOT need eigenstates!

We can compute energy with Loc. Wavefuncs.

# Locality of Wave Functions

Exponential localization (insulators):



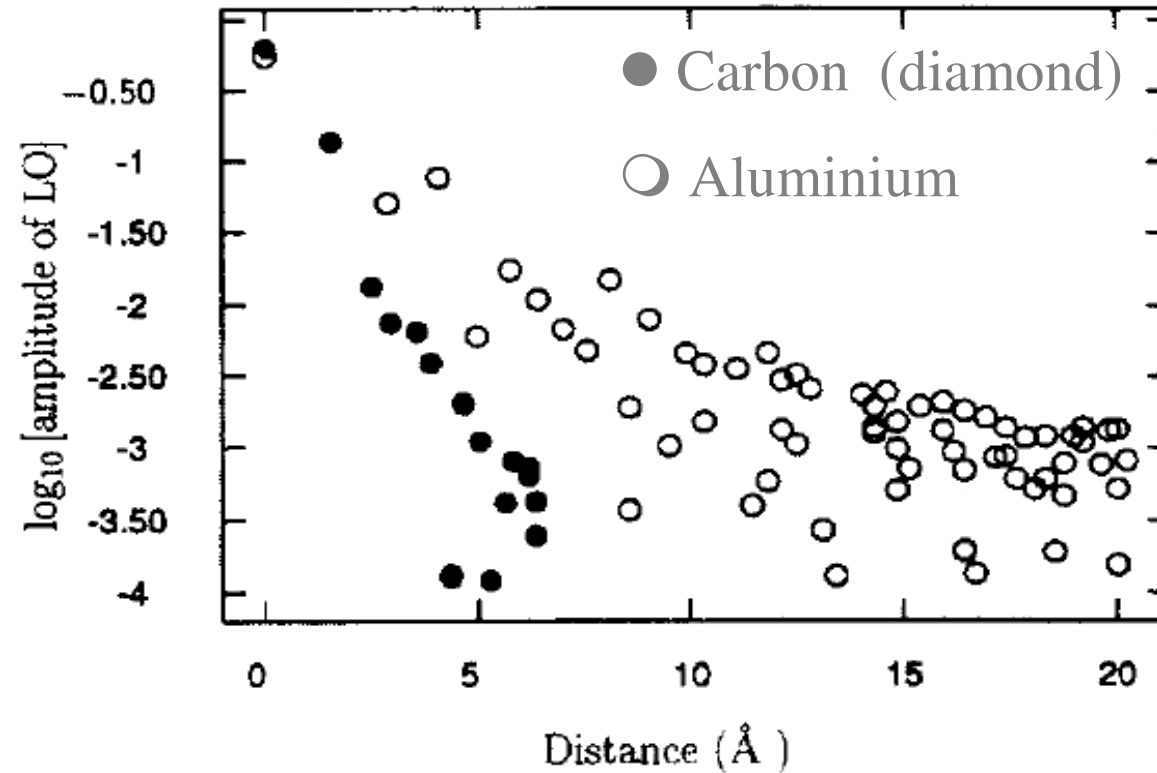
Wannier function in Carbon (diamond)

Drabold et al.



# Locality of Wave Functions

## Insulators vs Metals:



Goedecker & Teter, PRB 51, 9455 (1995)

# Order-N functionals

Kim, Mauri & Galli, PRB 52, 1640 (1995)

**Idea: derive a functional** that, when minimized, would lead to the correct ground state energy, but with the constraints would not need to be explicitly included

Constraints: orthonormalization of the one-electron wave function.

no computation of the inverse of the overlap matrix

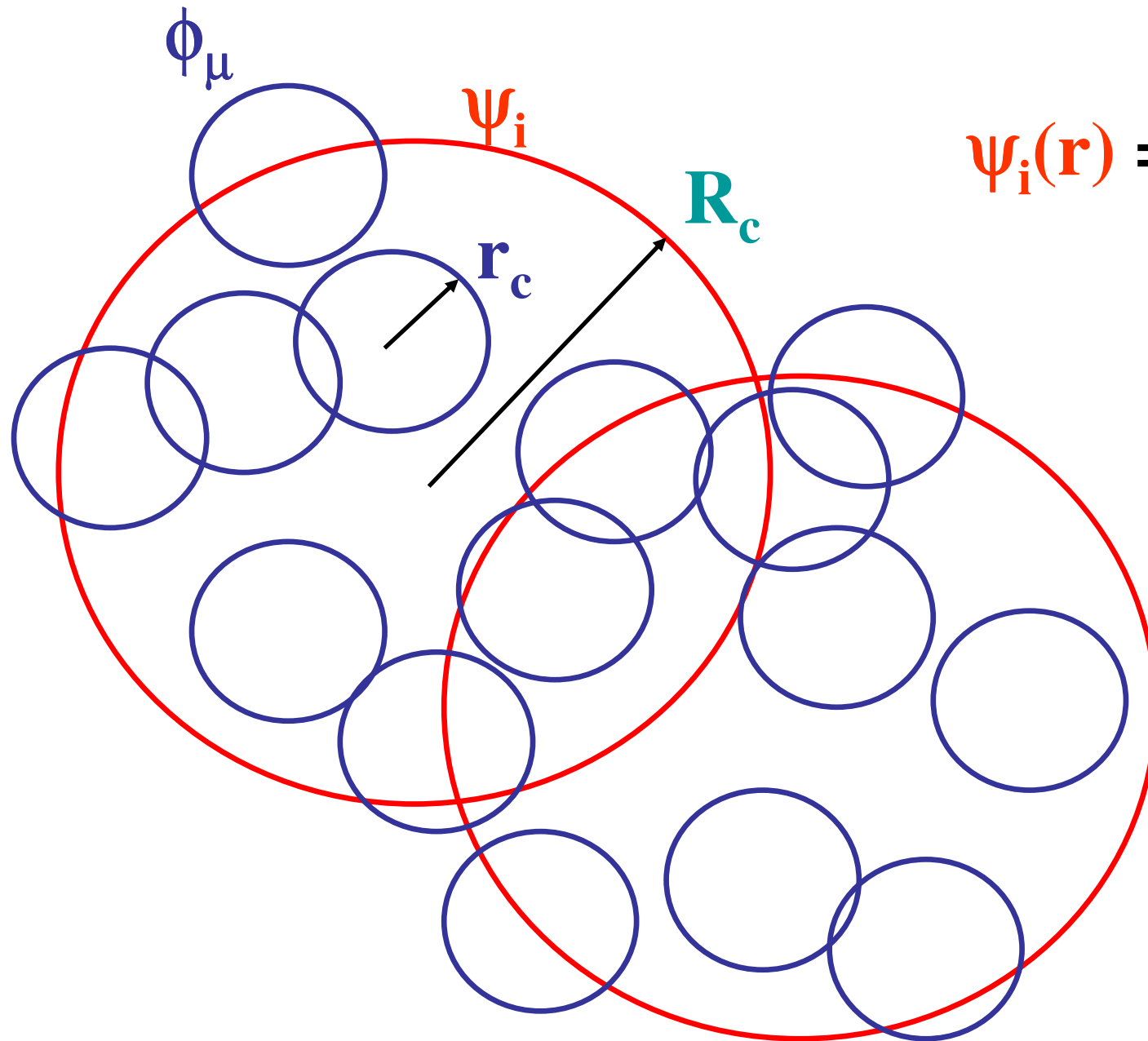
Functional of Ordejón-Mauri

$$\text{EOM} = \text{Trocc}[(2I-S) H] \quad \# \text{ states} = \# \text{ electron pairs}$$

Functional of Kim-Mauri-Galli

$$\text{EKMG} = \text{Trocc}+[(2I-S) (H-\eta S)] \quad \# \text{ states} > \# \text{ electron pairs}$$

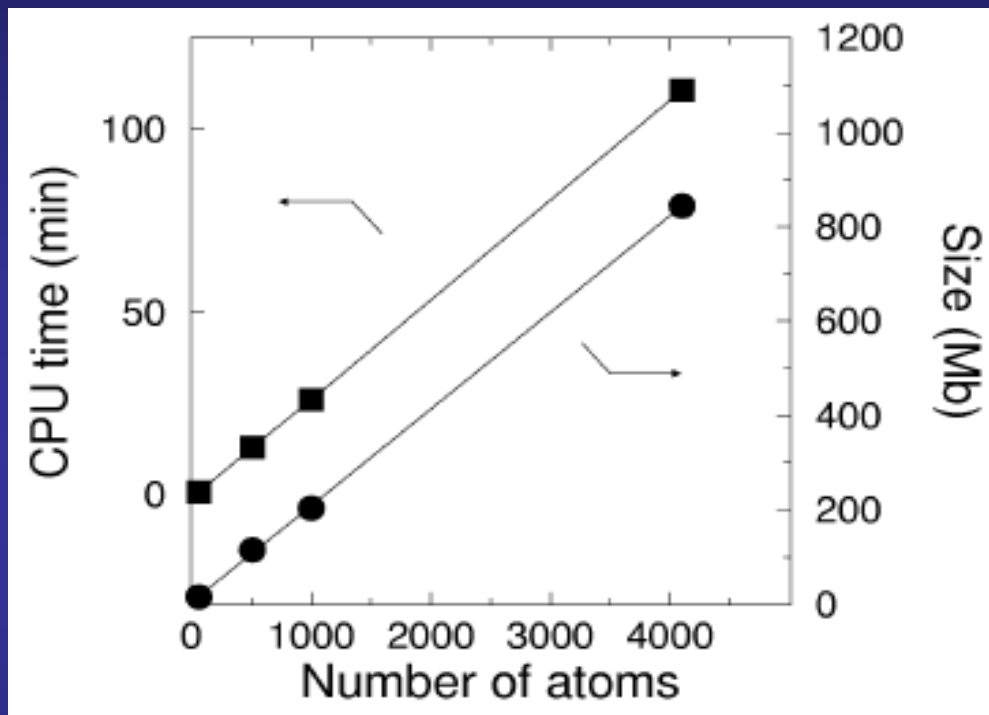
Minimization is done with respect a set of specially truncated wave functions, allowed to be non-zero only in a particular region of space



$$\psi_i(\mathbf{r}) = \sum_{\mu} \mathbf{c}_{i\mu} \phi_{\mu}(\mathbf{r})$$

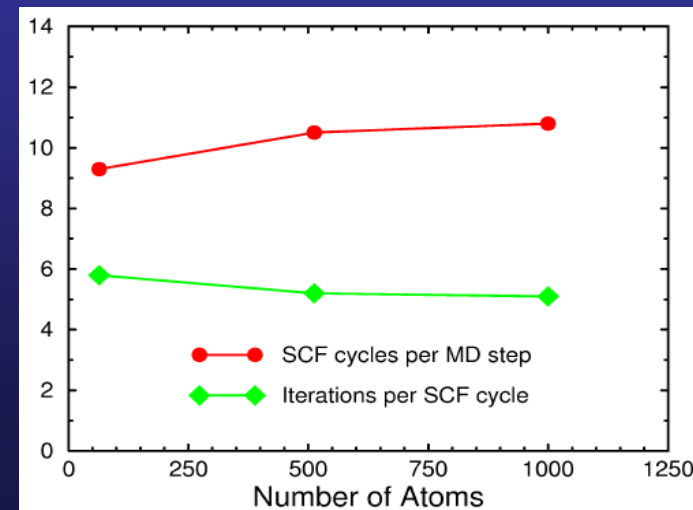
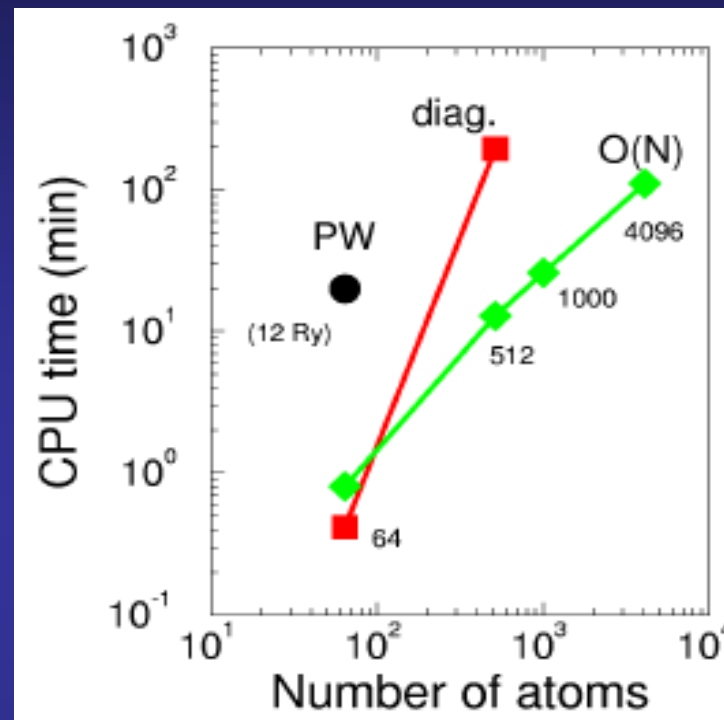
# Actual linear scaling

c-Si supercells, single- $\zeta$



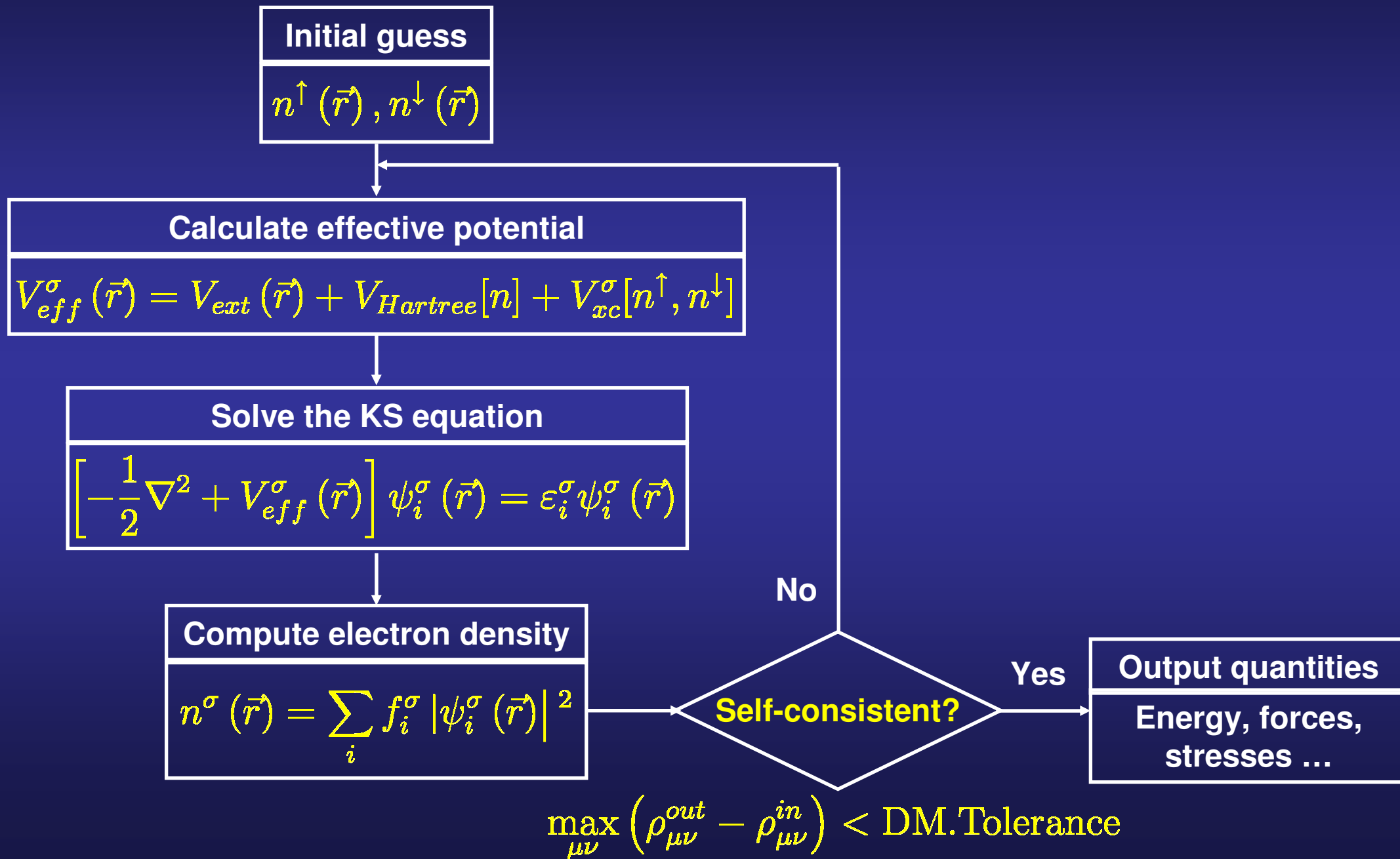
Single Pentium III 800 MHz. 1 Gb RAM

132.000 atoms in 64 nodes



# The Kohn-Sham equations must be solved self-consistently

## The potential (input) depends on the density (output)



**The Kohn-Sham total energy can be written as a sum of the band structure (BS) energy + ‘double-count’ corrections**

$$E^{BS} = \sum_i n_i \langle \psi_i | \hat{H} | \psi_i \rangle = \sum_{\mu\nu} H_{\mu\nu} \rho_{\nu\mu} = \text{Tr} (H \rho)$$

**After SCF**

**Eigenvectors of the Hamiltonian**

$$E^{KS} = \sum_{\mu\nu} H_{\mu\nu} \rho_{\nu\mu} - \underbrace{\frac{1}{2} \int V^H(\vec{r}) \rho(\vec{r}) d\vec{r} + \int [\epsilon^{xc}(\vec{r}) - V^{xc}(\vec{r})] \rho(\vec{r}) d\vec{r} + \sum_{IJ} \frac{Z_I Z_J}{R_{IJ}}}_{\text{Functionals of the electron density and atomic positions}}$$

**Functionals of the electron density and atomic positions**

# Kohn-Sham energy in SIESTA

$$\begin{aligned}
 E^{KS} = & \sum_{\mu\nu} T_{\mu\nu} \rho_{\nu\mu} && \text{Ekin} \\
 & + \sum_{\mu\nu} V_{\mu\nu}^{KB} \rho_{\nu\mu} && \text{Enl} \\
 & + \frac{1}{2} \sum_{IJ} U_{IJ}^{NA} (R_{IJ}) && \text{Ena} \\
 & + \sum_{I < J} \delta U_{IJ}^{local} (R_{IJ}) && \\
 & - \sum_I U_I^{local} && \text{Eions} \\
 & + \int V^{NA}(\vec{r}) \delta \rho(\vec{r}) d\vec{r} && \text{DEna} \\
 & + \frac{1}{2} \int \delta V_H(\vec{r}) \delta \rho(\vec{r}) d\vec{r} && \text{DUscf} \\
 & + \int \varepsilon^{xc}(\vec{r}) \rho(\vec{r}) d\vec{r} && \text{Exc}
 \end{aligned}$$

```

siesta: Program's energy decomposition (eV):
siesta: Eions      =      380.802124
siesta: Ena        =      114.848182
siesta: Ekin       =       81.633888
siesta: Enl        =       29.327240
siesta: DEna       =        4.386897
siesta: DUscf      =        0.250143
siesta: DUext      =        0.000000
siesta: Exc        =      -65.086299
siesta: eta*DQ     =        0.000000
siesta: Emadel     =        0.000000
siesta: Ekinion    =        0.000000
siesta: Eharris    =     -215.442072
siesta: Etot       =     -215.442072
siesta: FreeEng    =     -215.442072
    
```

+ Sum extra terms if a net charge (Emadel), an external electric field (DUext),  
Order-N solver (eta\*DQ) are used, or if the nuclei are moving (Ekinion)

# Atomic forces and stresses obtained by direct differentiation of the energy expression

$$\vec{F}_I = -\frac{\partial E^{KS}}{\partial \vec{R}_I}$$

$$\sigma_{\alpha\beta} = \frac{\partial E^{KS}}{\partial \epsilon_{\alpha\beta}} \quad \epsilon_{\alpha\beta} \equiv \text{strain tensor}$$

“One piece of energy  $\Rightarrow$  one piece of force and stress”

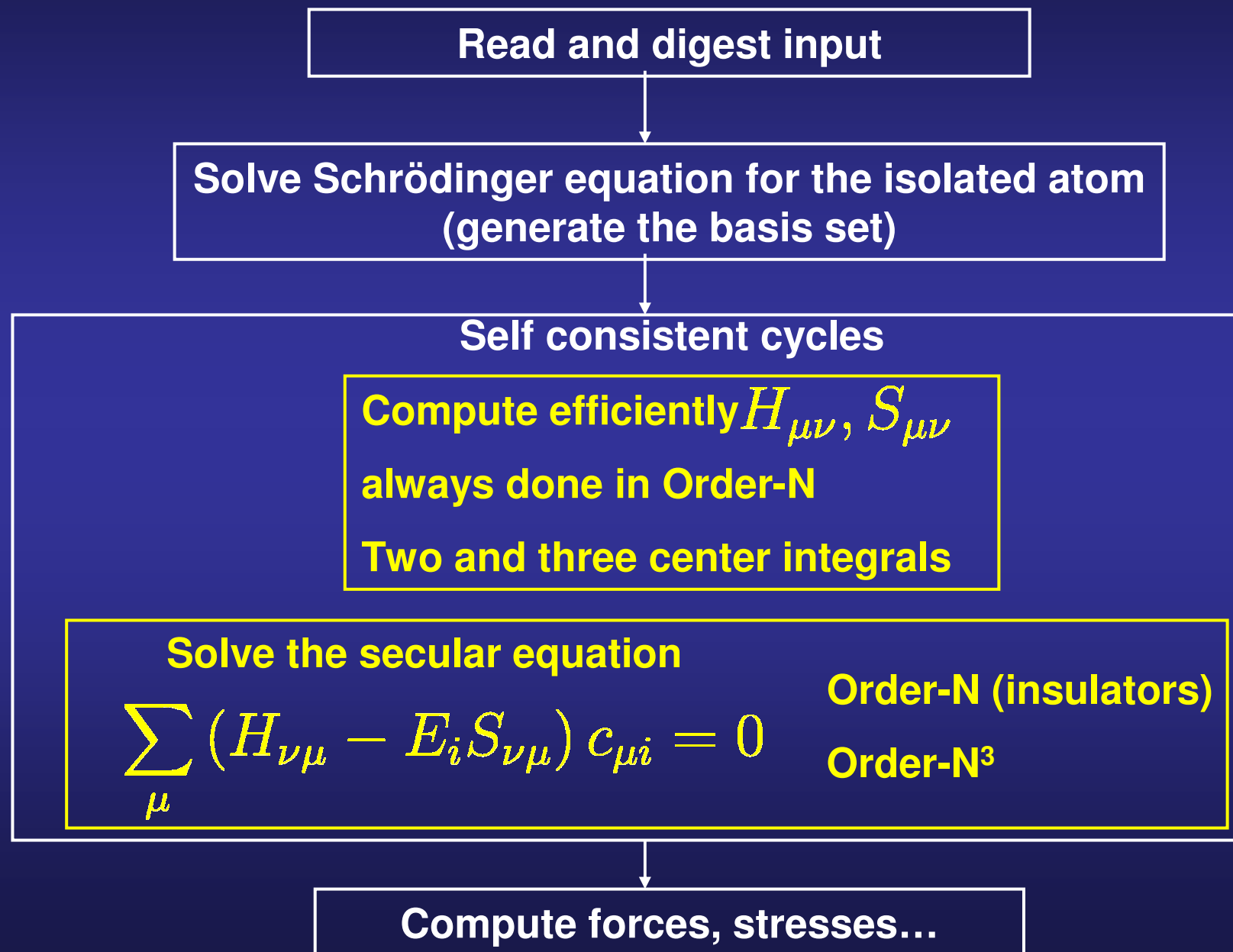
Calculated as the **analytical derivatives** of the energy

**Pulay corrections**, related with the dependency of the basis set on atomic positions, **automatically included**

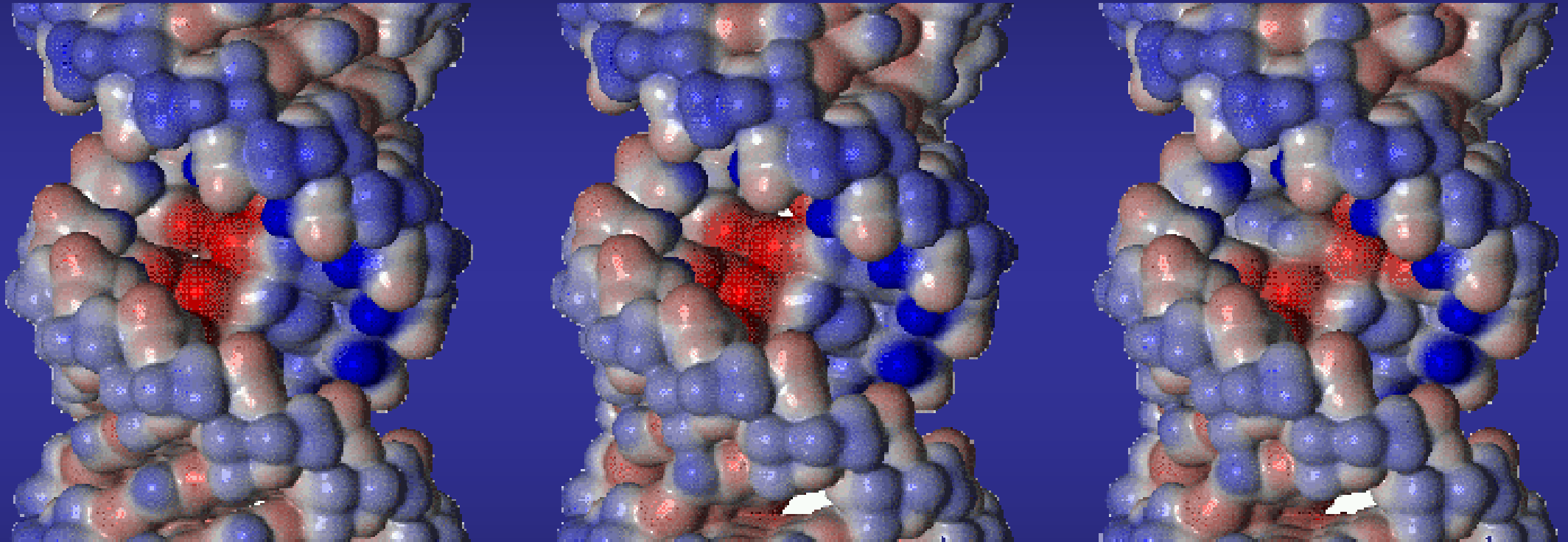
Calculated **only** in the **last self-consistent step**



# Recap: schematic flowchart of SIESTA



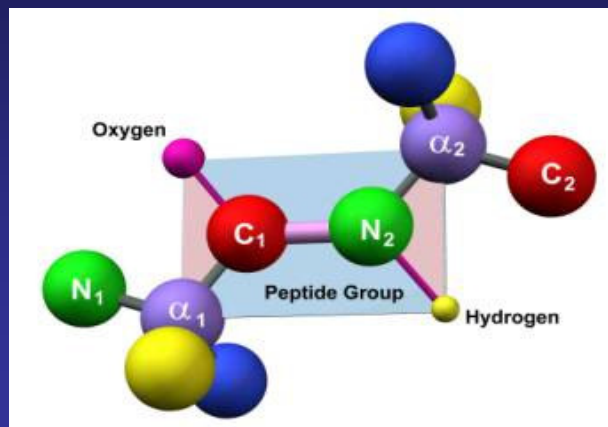
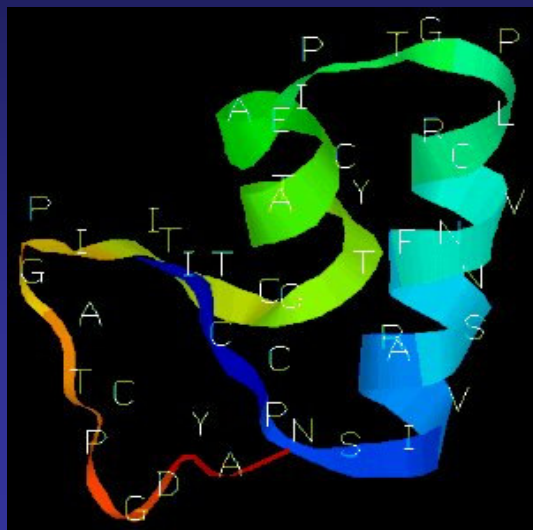
**The efficiency of Siesta allows the treatment of systems with an unprecedented number of atoms in the simulation box**



**Electronic structure of the double helix of DNA**

**715 atoms in a work station Digital Alpha**

# Deformation of the charge density in the peptidic bond of crambine



46 aminoacids; 642 atoms



PERGAMON

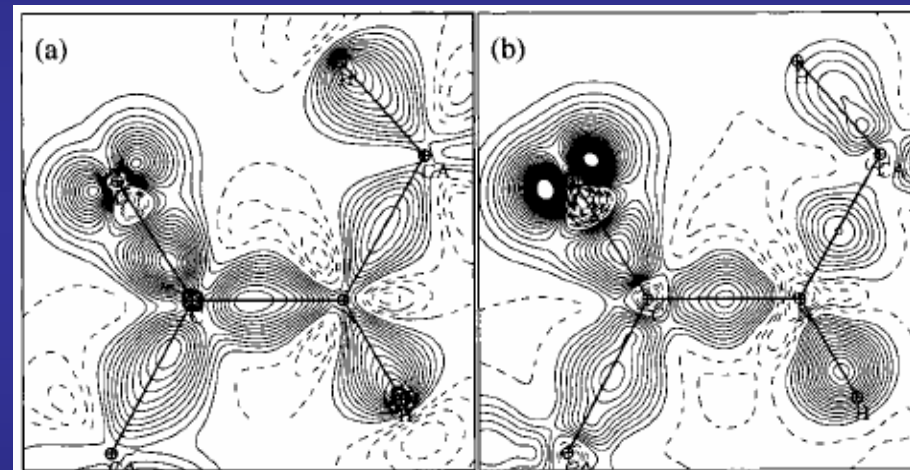
Solid State Communications 116 (2000) 395–400

solid  
state  
communications

[www.elsevier.com/locate/ssc](http://www.elsevier.com/locate/ssc)

Electron density in the peptide bonds of crambin

M.V. Fernández-Serra<sup>a,\*</sup>, J. Junquera<sup>a</sup>, C. Jelsch<sup>b</sup>, C. Lecomte<sup>b</sup>, E. Artacho<sup>c</sup>



X-ray diffraction

DFT (Siesta)

Average of all the peptidic bonds

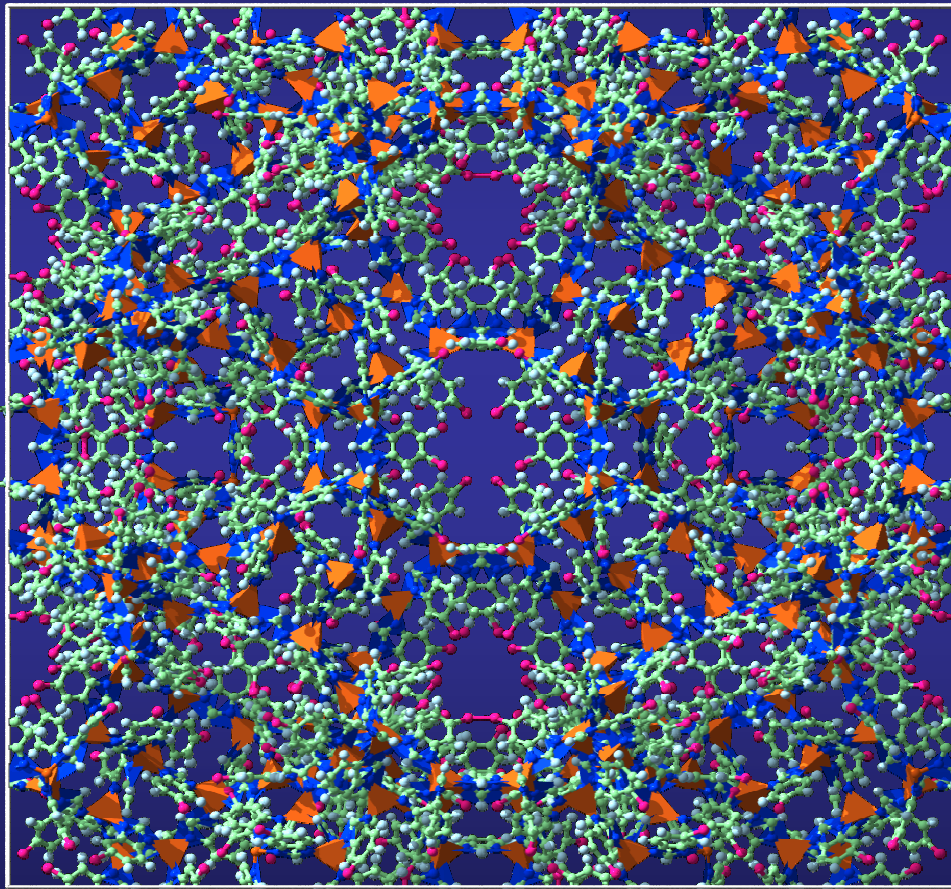
General results in good agreement with X-ray analysis

DZP basis set is accurate enough for the description

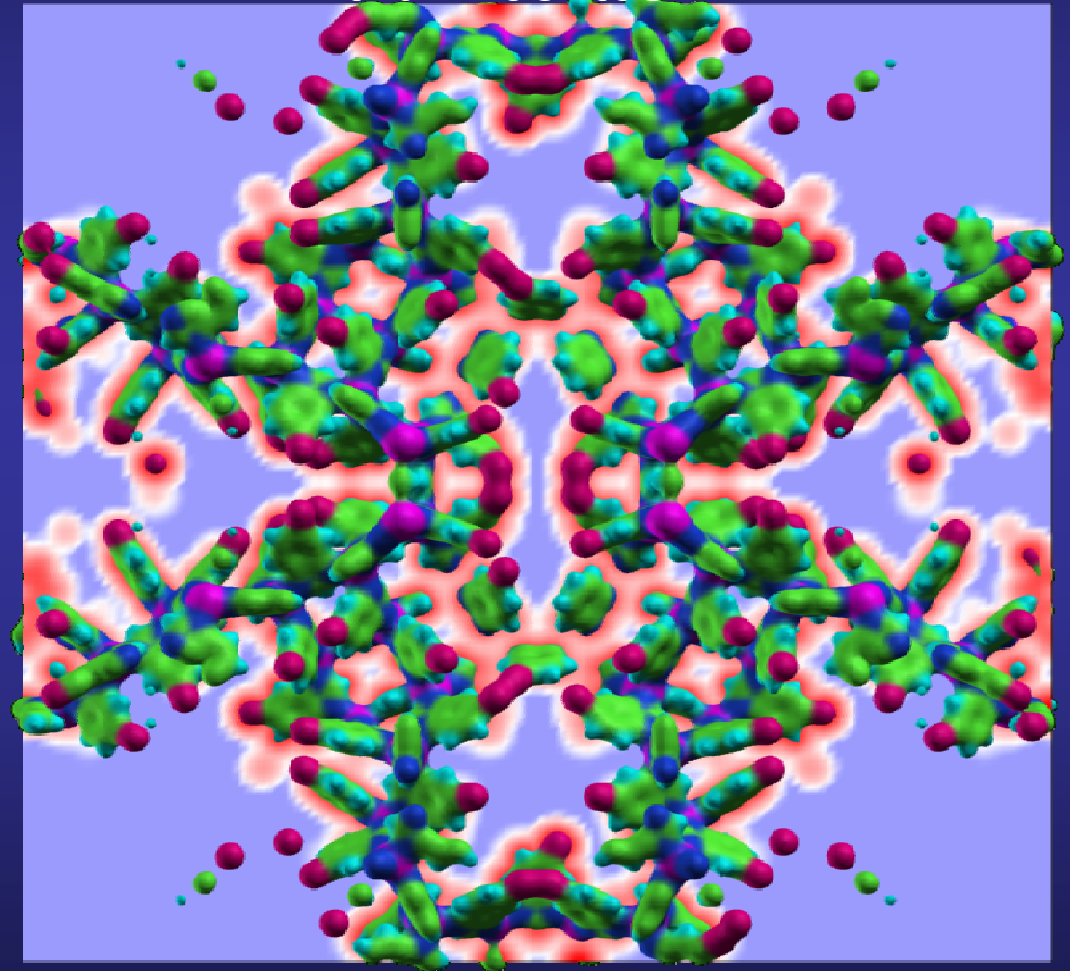
# Large/complex materials: Metal Organic Frameworks

B.O. Cankurtaran, J.D. Gale & M.J. Ford, J. Phys. Cond. Mat., 20, 294208 (2008)

Large space of empty space inside  
for gas adsorption



Lighter and easy to functionalize  
than zeolites



Largest calculation to date with SIESTA:  
524,288 atoms -> 2,000,000 basis fns

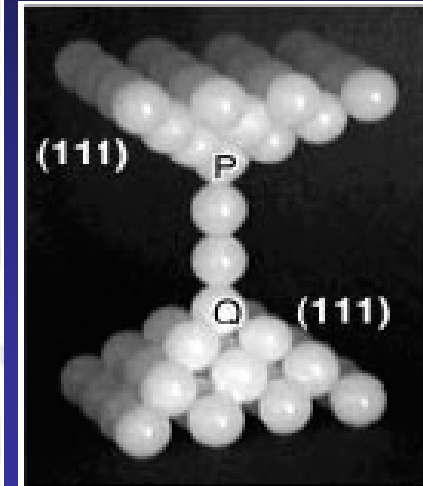
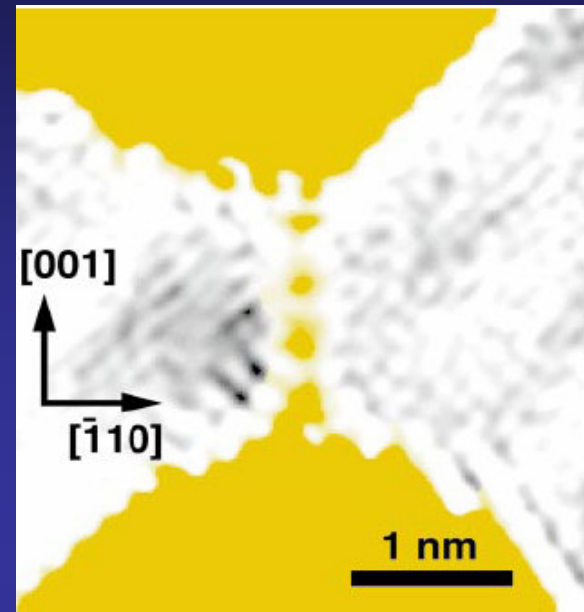


# Structure of monoatomic gold wires

NATURE | VOL 395 | 22 OCTOBER 1998 | www.nature.com

## Quantized conductance through individual rows of suspended gold atoms

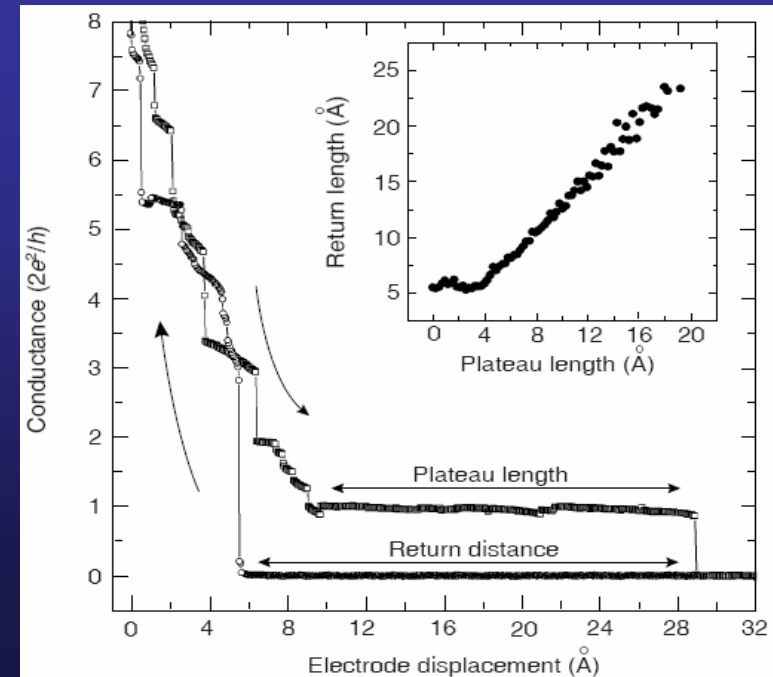
Hideaki Ohnishi\*, Yukihiro Kondo\* & Kunio Takayanagi\*\*



NATURE | VOL 395 | 22 OCTOBER 1998 | www.nature.com

## Formation and manipulation of a metallic wire of single gold atoms

A. I. Yanson\*, G. Rubio Bollinger†, H. E. van den Brom\*, N. Agraït† & J. M. van Ruitenbeek\*



# Structure of monoatomic gold wires

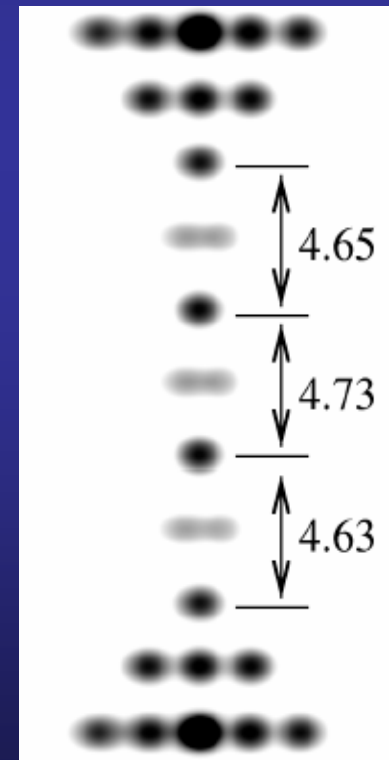
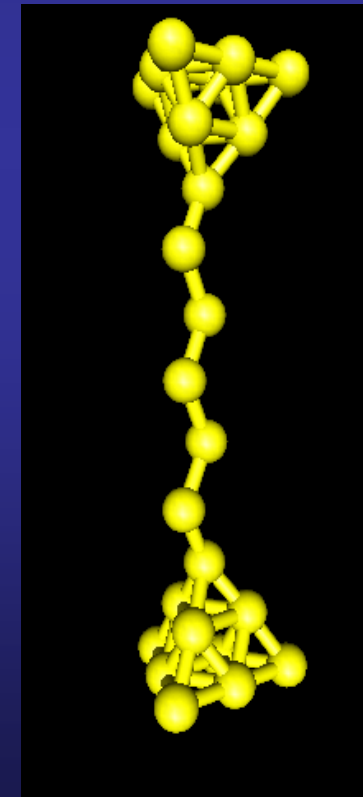
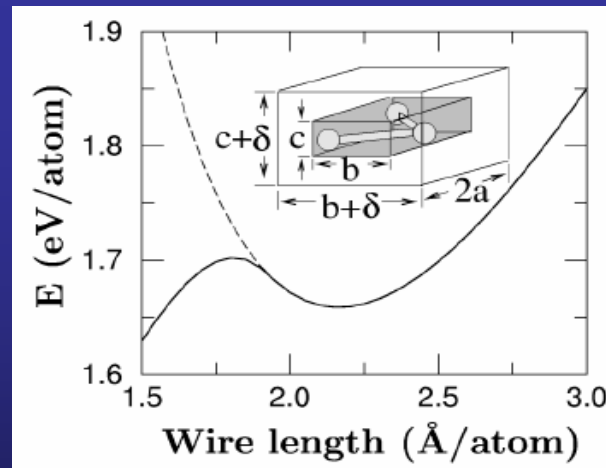
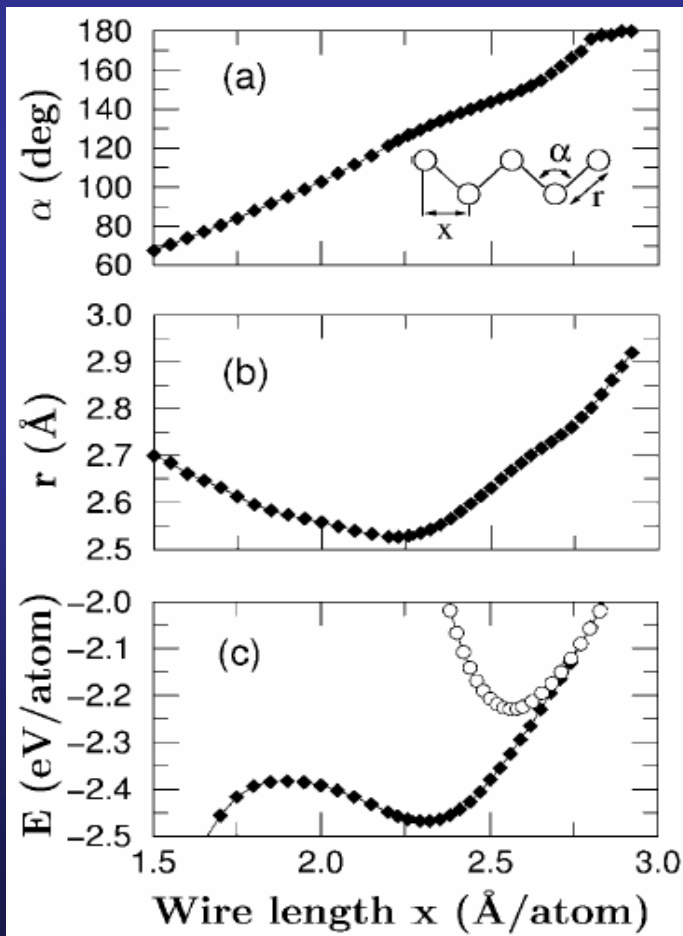
VOLUME 83, NUMBER 19

PHYSICAL REVIEW LETTERS

8 NOVEMBER 1999

## Stiff Monoatomic Gold Wires with a Spinning Zigzag Geometry

Daniel Sánchez-Portal,<sup>1</sup> Emilio Artacho,<sup>2</sup> Javier Junquera,<sup>2</sup> Pablo Ordejón,<sup>3</sup> Alberto García,<sup>4</sup> and José M. Soler<sup>2</sup>



# Nanospintronics:

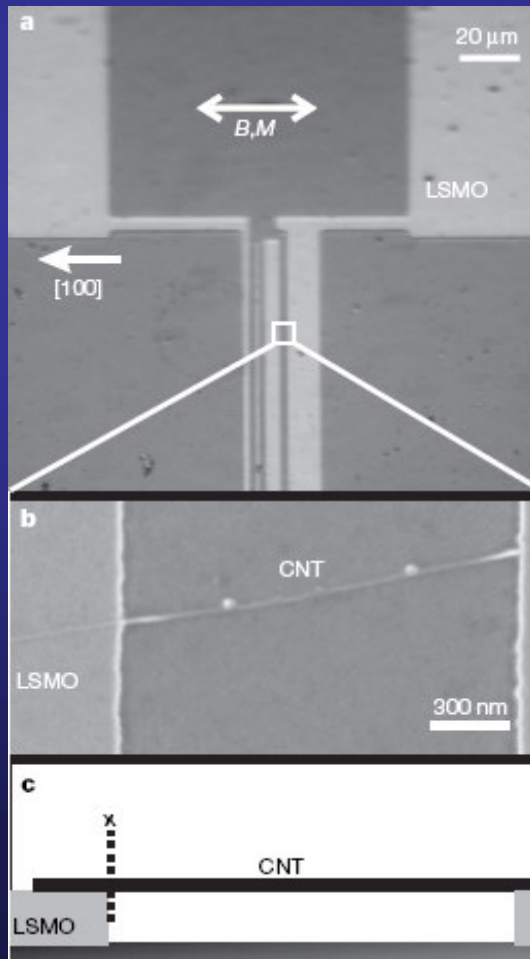
## Nanotubes to transport spin

nature

Vol 445 | 25 January 2007 | doi:10.1038/nature05507

### Transformation of spin information into large electrical signals using carbon nanotubes

Luis E. Hueso<sup>1†</sup>, José M. Pruneda<sup>2,3†</sup>, Valeria Ferrari<sup>4†</sup>, Gavin Burnell<sup>1†</sup>, José P. Valdés-Herrera<sup>1,5</sup>, Benjamin D. Simons<sup>4</sup>, Peter B. Littlewood<sup>4</sup>, Emilio Artacho<sup>2</sup>, Albert Fert<sup>6</sup> & Neil D. Mathur<sup>1</sup>



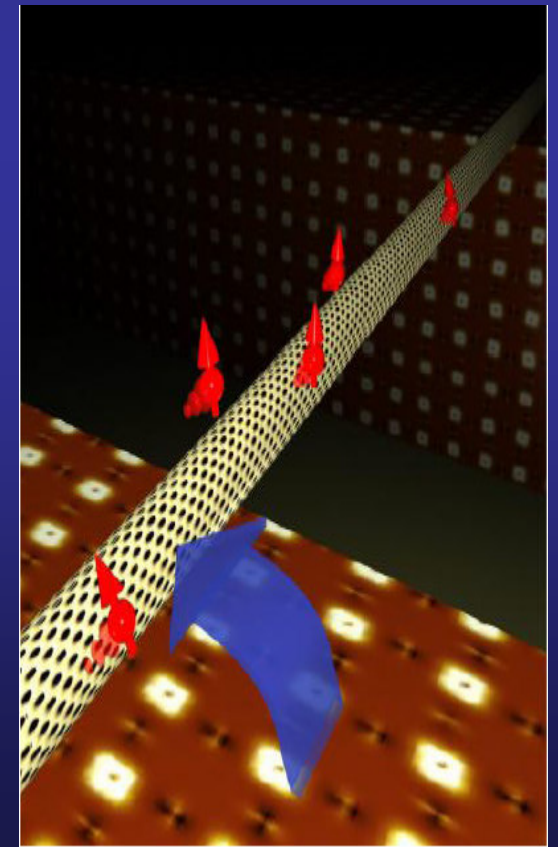
**Inject** spins in a nonmagnetic material

**Transport** this spin over distances of thousands of nm

**Detect** the spin at the other end of the device with a large enough electric signal

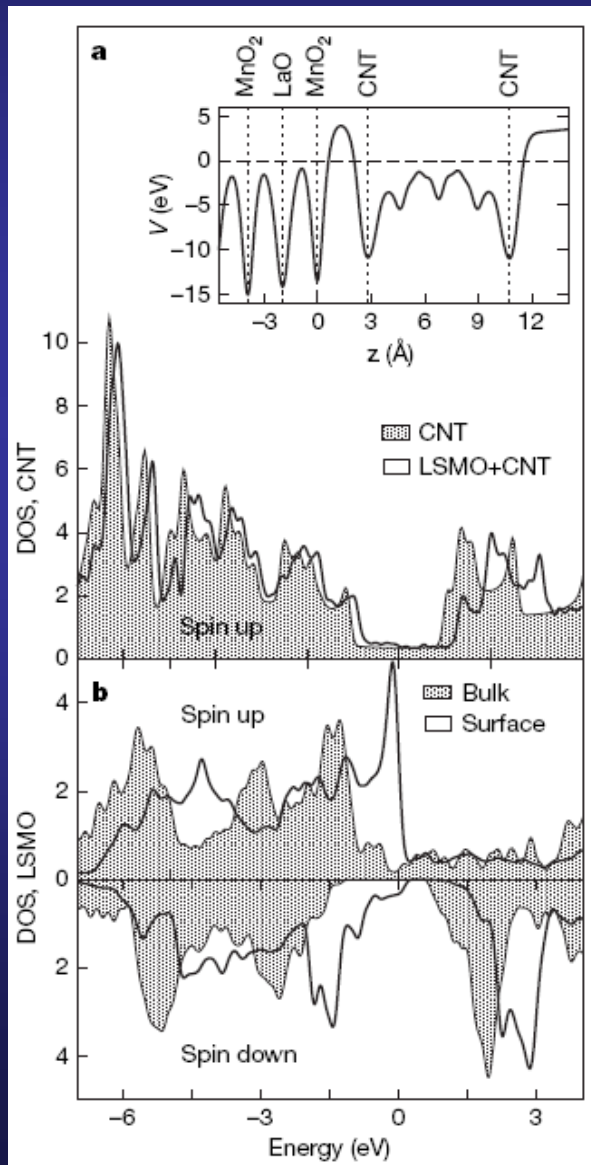
Magnetoresistance

$$MR = \frac{\Delta R}{R_P} \equiv \frac{R_{AP} - R_P}{R_P}$$



# Nanospintronics:

## Nanotubes to transport spin



nature

Vol 445 | 25 January 2007 | doi:10.1038/nature05507

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Energy barrier at the CNT/LSMO interface, but the ratio kinetic energy and the barrier width suggest a relatively high transmission probability

Carbon nanotube (CNT) not significantly altered when contacted by La<sub>0.7</sub>Sr<sub>0.3</sub>MnO<sub>3</sub> (LSMO)

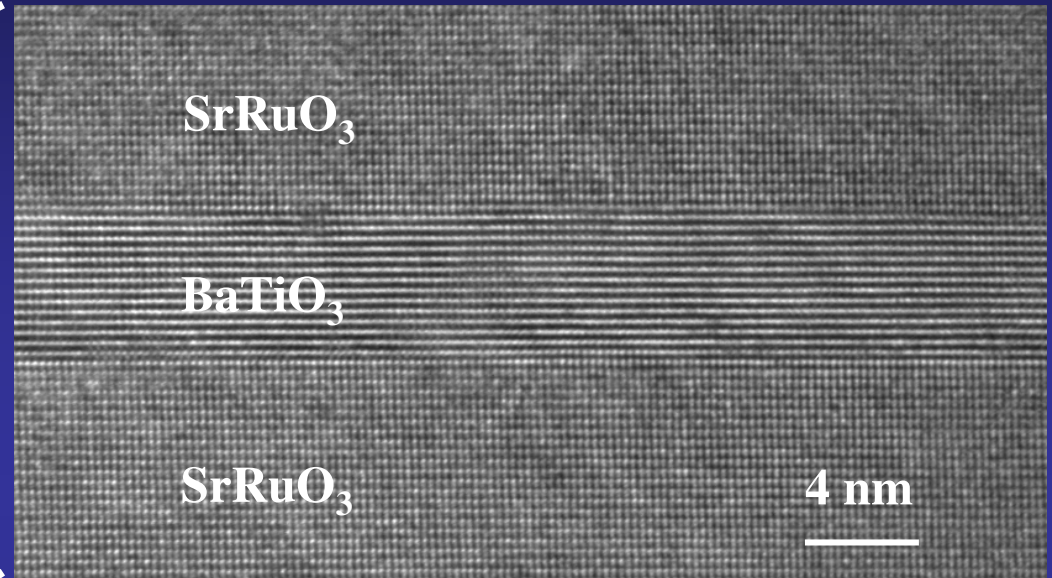
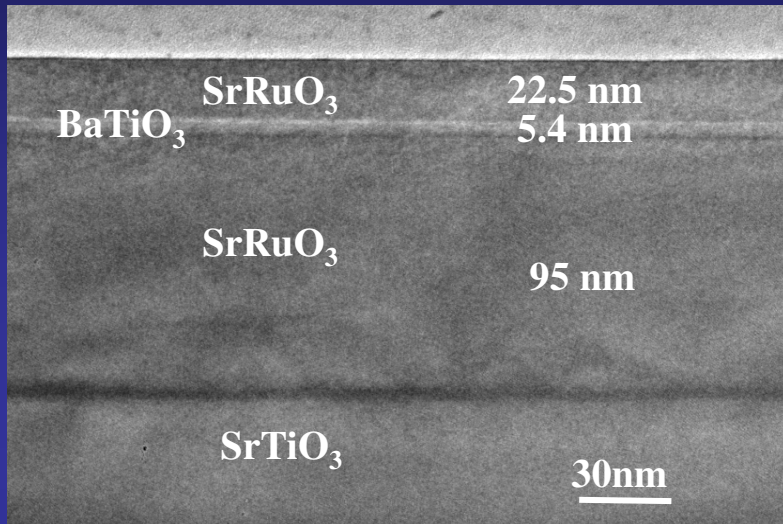
The LSMO surface is highly spin polarized

$$MR = \frac{\Delta R}{R_P} \equiv \frac{R_{AP} - R_P}{R_P} \quad \mathbf{61 \% \text{ at } 5 \text{ K}}$$

Previous devices limited to ~0.1-1 %



# First simulation on a realistic ferroelectric capacitor: existence of a critical thickness for ferroelectricity



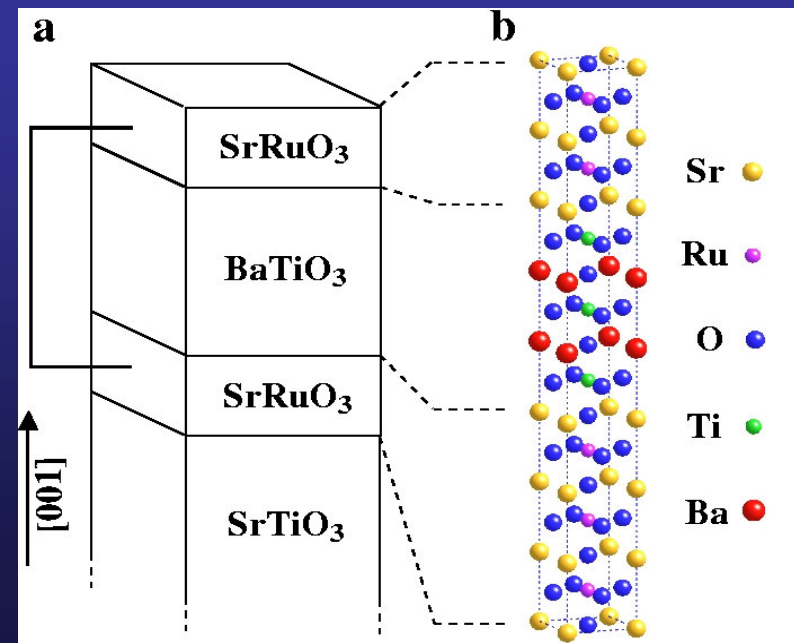
NATURE | VOL 422 | 3 APRIL 2003 | [www.nature.com/nature](http://www.nature.com/nature)

## Critical thickness for ferroelectricity in perovskite ultrathin films

Javier Junquera & Philippe Ghosez

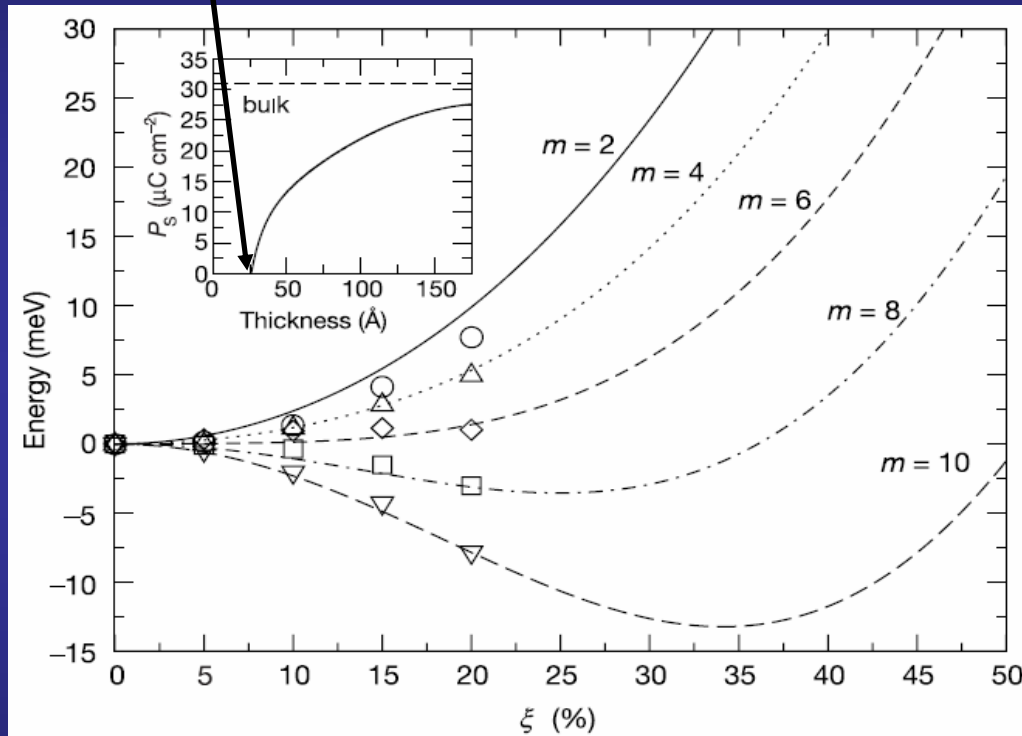
Département de Physique, Université de Liège, Bâtiment B-5, B-4000  
Sart-Tilman, Belgium

253 citas



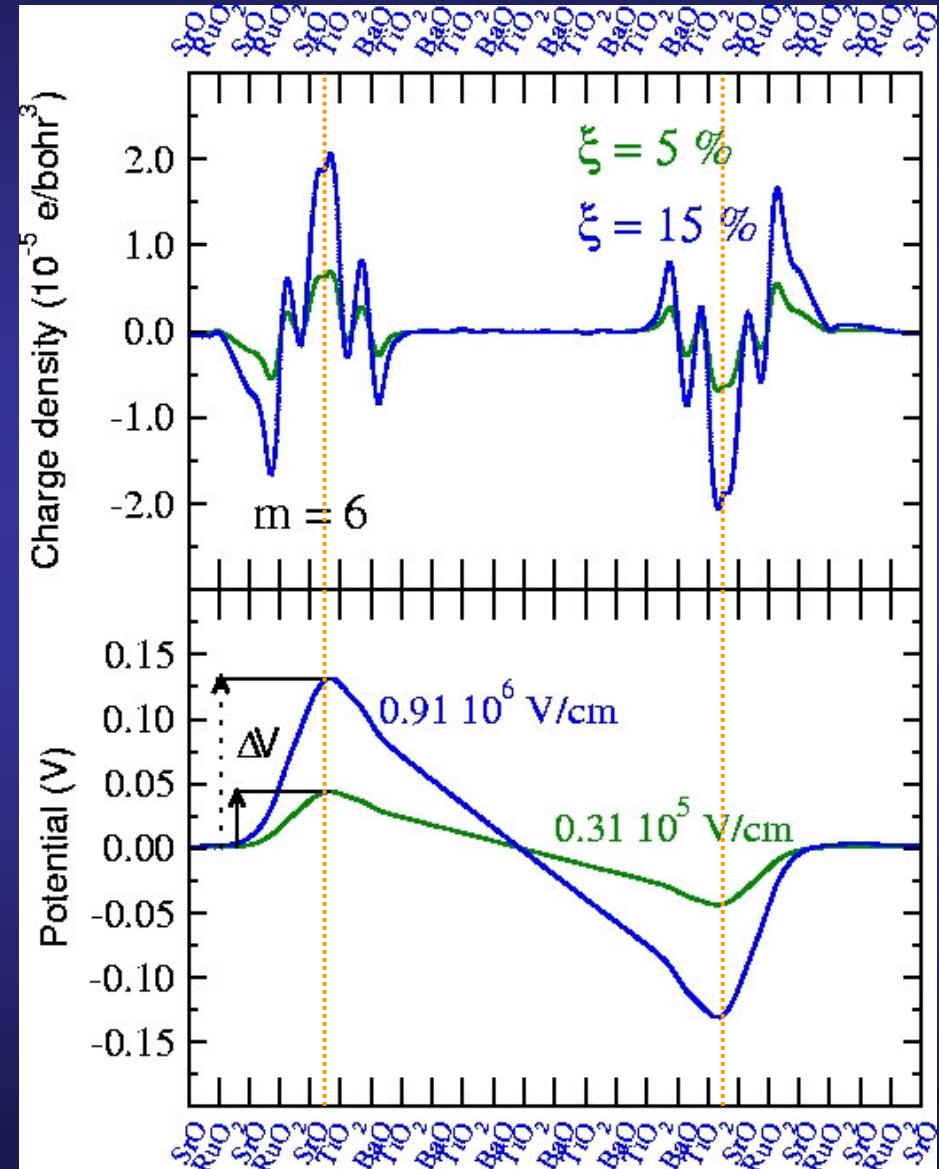
# The critical thickness is due to the incomplete screening of polarization charge: the depolarizing field

$t_c \approx 24 \text{ \AA}$  (6 celdas unidad)



Depolarizing field:

- Directly proportional to the polarization.
- Inversely proportional to the thickness.



# **SIESTA availability**

- **Free to academics**
- **Free to Government labs engaged in academic & ethical research**
- **Small charge to industry**
- **All get source code**

# SIESTA current status

- **New versions about to be released**
  - Massively parallel re-write (Barcelona Supercomputer Centre)
  - Van der Waals version
  - TranSIESTA included (PRB, 65, 165401 (2002))
  - LDA + U
  - Z-matrices
  - Broyden optimisation
  - SIESTA as a subroutine / CamposASE (<http://www.fysik.dtu.dk/campos/ASE>)
  - Many enhancements / new properties

# SIESTA next version...

- **New order N solvers**
  - Divide and conquer
    - (*J. Phys. Cond. Matter*, 20, 294208 (2008))
  - Li, Nunes, Vanderbilt
  - Density matrix purification
- **Filterets**
- **Time Dependent Density Functional Theory (TDDFT)**
- **Spin-orbit**
- **Response of periodic dielectrics to electric field**
- **Calculation of optical properties**
- **Simulation of STM images**

# Example of input

## H<sub>2</sub>O molecule: example of a very simple input file

Go to the directory where the exercise of the H<sub>2</sub>O molecule is included

Inspect the input file, h2o.fdf

Examine in detail the different input variables, more information at

<http://www.icmab.es/siesta> and follow the link Documentations, Manual

```
SystemName          Water molecule
SystemLabel         h2o
NumberOfAtoms       3
NumberOfSpecies      2

%block ChemicalSpeciesLabel
  1  8  0      # Species index, atomic number, species label
  2  1  H
%endblock ChemicalSpeciesLabel

AtomicCoordinatesFormat Ang
%block AtomicCoordinatesAndAtomicSpecies
  0.000  0.000  0.000  1
  0.757  0.586  0.000  2
 -0.757  0.586  0.000  2
%endblock AtomicCoordinatesAndAtomicSpecies
```

Number of different  
species and atoms  
present in the unit cell

List of different species

Position of the atoms

Example of a first-principles simulation: no input from experiment



# Input of example

## MgO an ionic solid that crystallizes in the rocksalt structure

Go to the directory where the exercise of the bands of MgO is included

Inspect the input file, MgO.fdf

```
SystemName Magnesium Oxide Crystal
SystemLabel          MgO

NumberOfAtoms        2
NumberOfSpecies       2

%block Chemical_Species_Label
  1  12  Mg
  2   8   0
%endblock Chemical_Species_Label

LatticeConstant      4.117 Ang
%block LatticeVectors
  0.000  0.500  0.500
  0.500  0.000  0.500
  0.500  0.500  0.000
%endblock LatticeVectors

AtomicCoordinatesFormat Fractional
%block AtomicCoordinatesAndAtomicSpecies
  0.000  0.000  0.000  1
  0.500  0.500  0.500  2
%endblock AtomicCoordinatesAndAtomicSpecies

%block kgrid_Monkhorst_Pack
  6  0  0  0.5
  0  6  0  0.5
  0  0  6  0.5
%endblock kgrid_Monkhorst_Pack
```

More information at the [Siesta web page](http://www.icmab.es/siesta)  
<http://www.icmab.es/siesta> and follow the  
link Documentations, Manual

The equilibrium lattice constant within  
LDA has been computed for you...

Rocksalt structure:

FCC lattice

+ a basis of two atoms

Sampling in k in the first Brillouin zone  
to achieve self-consistency

# Recap on SIESTA

- First-principles method and Fortran-90 implementation code
- Based on Density Functional Theory
- Norm-conserving pseudopotential
- **Strictly localized Numerical Atomic Orbital as the basis set**
- **Matrix elements of the Hamiltonian and Overlap matrices always computed in Order-N.**
- **Solution of the Hamiltonian, in Order-N or standard diagonalization.**
- Freely available for the academic community

More information at: <http://www.uam.es/siesta>

<http://personales.unican.es/junqueraj>